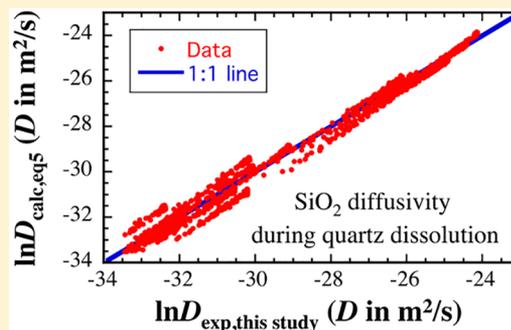


# Kinetics of Quartz Dissolution in Natural Silicate Melts and Dependence of SiO<sub>2</sub> Diffusivity on Melt Composition

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## S Supporting Information

**ABSTRACT:** Quartz is a major mineral in silicic rocks, and a common phenocryst in rhyolite. An understanding of the kinetics of quartz dissolution and growth may provide insight into magma crystallization and constraints on magma dynamics and cooling rates. We have carried out quartz dissolution experiments in rhyolitic (0.1 wt % H<sub>2</sub>O, ~73 wt % SiO<sub>2</sub>) and basaltic (~0.35 wt % H<sub>2</sub>O, and ~50 wt % SiO<sub>2</sub>) melts at 1300–1600 °C and 0.5 GPa using piston cylinder apparatus. The experiments constrain the interface melt compositions at quartz saturation up to 1600 °C, which depend on whether the initial melt is rhyolite or basalt. The data on silica concentration at quartz saturation in each melt are modeled and will be important for future improvement of thermodynamic models of silicate melts. In addition, the experiments provide data on SiO<sub>2</sub> diffusivity, which plays major roles in the kinetics and dynamics of various igneous processes, including magma mixing. SiO<sub>2</sub> diffusivity depends on melt composition, consistent with previous results. Combined with other data from our lab and literature, we show that  $\ln D_{\text{SiO}_2}$  decreases linearly with  $X$  (= Si + Al cation mole fraction) in rhyolitic to andesitic to basaltic melts, instead of just the SiO<sub>2</sub> concentration. The effect of H<sub>2</sub>O is also captured by  $X$  when H<sub>2</sub>O is included in the cation mole fraction calculation. Each SiO<sub>2</sub> diffusion profile during quartz dissolution can be fit well by assuming  $D_{\text{SiO}_2} = D_{X=1} e^{\beta(1-X)}$ . Using data from our experiments, SiO<sub>2</sub> diffusivity during quartz dissolution in rhyolitic, andesitic, to basaltic melts can be expressed as the following Arrhenius relation with compositional dependence,  $D_{\text{SiO}_2}^{\text{quartz dissolution}} = \exp(-14.168 + 2.758(1 - X) - [(35003 - 38829(1 - X))/T])$ , where  $D_{\text{SiO}_2}$  is in m<sup>2</sup>/s and  $T$  is in K. The 1 $\sigma$  standard deviation and maximum deviation of the above equation in predicting  $\ln D_{\text{SiO}_2}$  are 0.32 (or 0.14 log<sub>10</sub>  $D$  units) and 0.95 (or 0.41 log<sub>10</sub>  $D$  units). Because SiO<sub>2</sub> diffusivity depends on SiO<sub>2</sub> concentration, no theory is currently available to predict diffusive quartz dissolution rate. We develop a method by adopting the formulation for the case of constant  $D$  but replacing the constant  $D$  by an effective  $D$  ( $D_{\text{eff}}$ ),  $L = 2\alpha(D_{\text{eff}}t)^{1/2}$ , where  $\alpha$  is solved from the composition of the dissolving crystal, the interface melt and the far-field melt. Using experimental data, the effective diffusivity during mineral dissolution may be related to diffusivity in the farfield ( $D_{\text{farfield}}$ ) and interface ( $D_{\text{interface}}$ ) melts as follows:  $\ln(D_{\text{eff}}/D_{\text{farfield}}) = (0.6996 + 0.0327Y)Y$ , where  $Y = \ln(D_{\text{interface}}/D_{\text{farfield}})$ . The method is applied successfully to treat diffusive and convective quartz dissolution rates.



**KEYWORDS:** Quartz dissolution kinetics, SiO<sub>2</sub> diffusion, Mineral dissolution rate with variable diffusivity, SiO<sub>2</sub> concentration at quartz saturation, Quartz-melt equilibrium

## 1. INTRODUCTION

Quartz is a major mineral in silicic igneous rocks. An understanding of the kinetic interaction between quartz and silicate melts can help model quartz growth and dissolution in melts. In addition, because quartz is a simple stoichiometric mineral, there is no compositional variability for modeling quartz growth and dissolution. Hence, modeling quartz dissolution and growth rates in silicate melts may be more accurate than that of other minerals, and the results may be applied to evaluate magma dynamics and cooling rates.

There have been many studies on different aspects of quartz dissolution in various silicate melts. For example, Watson<sup>1</sup> studied quartz dissolution in a basaltic melt and examined the

effect of convection. Zhang et al.<sup>2</sup> conducted an experiment of quartz dissolution in an andesitic melt. Liang<sup>3</sup> investigated the diffusivity matrix during quartz dissolution in a ternary CAS (CaO–Al<sub>2</sub>O<sub>3</sub>–SiO<sub>2</sub>) melt system. Shaw<sup>4–7</sup> and Acosta-Vigil<sup>8</sup> focused on the interface reaction between quartz, and natural and various synthetic melts. These studies have provided insight into different aspects of quartz dissolution kinetics. However, there is a need to quantify quartz dissolution and

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Table 1. Compositions of Quartz and Rhyolitic, Andesitic, and Basaltic Glasses<sup>a,b</sup>

oxides	mineral		rhyolitic glass			andesite	basalt
	quartz	NCO	CIT	KS	H6a	LML <sup>c</sup>	JDF <sup>d</sup>
SiO <sub>2</sub>	99.78	72.89	76.3	76.46	74.67	56.5	49.9
SnO <sub>2</sub>	nd	nd	nd	nd	nd	nd	nd
TiO <sub>2</sub>	0.01	0.22	0.05	0.06	0.26	1.24	1.83
Al <sub>2</sub> O <sub>3</sub>	0.12	14.23	12.98	12.43	13.44	18.0	13.53
FeO <sub>t</sub>	0.11	1.93	0.91	1.02	1.72	6.71	12.93
MnO	0.01	0.06	nd	nd	nd	0.13	0.22
MgO	nd	0.18	0.02	0.03	0.27	3.96	6.81
CaO	0.02	0.86	0.41	0.49	1.22	7.73	10.81
Na <sub>2</sub> O	nd	4.73	4.06	4.25	3.92	3.75	2.65
K <sub>2</sub> O	0.01	4.24	5.28	5.01	4.49	1.70	0.17
H <sub>2</sub> O	nd	0.1	0.1	0.9	5.9	0.02	0.32
dry total	100.06	99.35	100.01	99.75	99.99	99.74	98.85

<sup>a</sup>nd means not determined. <sup>b</sup>SiO<sub>2</sub> to K<sub>2</sub>O concentrations in KS and H6a are given on an anhydrous basis for easy comparison. <sup>c</sup>Composition of LML andesite is from Zhang et al. (1989) with H<sub>2</sub>O concentration from FTIR measurement. <sup>d</sup>Water content in JDF is given by averaging the range 0.25–0.4 wt % H<sub>2</sub>O (Chen and Zhang, 2008)

Table 2. Summary of Experimental Conditions<sup>a,d</sup>

type	exp#	melt used	<i>T</i> (°C)	duration (s)	<i>L</i> <sub>melt</sub> (μm)	<i>L</i> <sub>mineral</sub> (μm)
quartz dissolution in rhyolite	QzDisRh103	NCO	1303	172811	12.5	11.6
	QzDisRh111		1304	87060	14.8	13.7
	QzDisRh112		1305	43219	9.0	8.3
	QzDisRh115		1293	346258	24.1	22.3
	QzDisRh201		1294	21583	6.4	5.9
	QzDisRh203		1292	86397	12.9	11.9
	QzDisRh105		1408	86433	36.5	33.8
	QzDisRh113		1414	44970	20.9	19.3
	QzDisRh114		1403	21627	21.2	19.6
	QzDisRh102		1505	14386	17.2	15.9
	QzDisRh104		1501	86432	50.3	46.5
	QzDisRh106		1600	14435	43.8	40.5
	quartz dissolution in andesite <sup>b</sup>	234	LML	1300	3600	6.5
quartz dissolution in basalt	QzDisBa101	JDF	1293	3667	29.6	32.3
	QzDisBa102		1306	918	17.6	19.2
	QzDisBa103		1304	257	8.6	9.4
	QzDisBa110		1394	1232	39.1	42.6
	QzDisBa111		1400	318	17.4	19.0
	QzDisBa104		1508	928	43.3	47.2
	QzDisBa107		1576	322	34.9	38.1
	cassiterite dissolution in rhyolite <sup>c</sup>	CassDis8	H6a	950	239	1.5
CassDis9			750	634	0.08	0.10
CassDis6			850	299	0.31	0.09
CassDis12			850	1519	0.65	0.21
CassDis10		KS	900	18026	0.67	0.22
CassDis3			1000	18092	3.4	1.1
CassDis1			1100	1856	2.6	0.9
CassDis11		CIT	1000	18000	2.3	0.8
CassDis13			1100	3633	0.27	0.09

<sup>a</sup>All experiments were conducted at 0.5 GPa pressure condition. Reported experimental temperature *T* is the corrected temperature using the calibration of ref 38. One micrometer = 10<sup>-6</sup> m. <sup>b</sup>From ref 2 with *L*<sub>melt</sub> and *L*<sub>mineral</sub> from our new fit. <sup>c</sup>From ref 17. <sup>d</sup>Melt growth distance *L*<sub>melt</sub> is calculated from SiO<sub>2</sub> concentration profiles using mass balance for quartz dissolution and cassiterite dissolution experiments; mineral dissolution distance (*L*<sub>mineral</sub>) is then calculated by using  $L_{\text{mineral}} = L_{\text{melt}} \rho_{\text{melt}} / \rho_{\text{mineral}}$ .  $\beta$ -quartz density of 2.54 g/cc; NCO, KS, and CIT densities of 2.35 g/cc; H6a density of 2.28 g/cc; LML density of 2.60 g/cc; JDF basalt glass density of 2.77 g/cc; cassiterite density of 6.99 g/cc.

growth rate and SiO<sub>2</sub> diffusivity in different silicate melts. In addition, more data on the interface melt composition during quartz dissolution (i.e., at quartz saturation) can be used to improve thermodynamic models of silicate melts, such as the MELTS software.<sup>9–13</sup> Therefore, a systematic investigation of

the kinetics of quartz dissolution in different silicate melts is important to a number of geological problems.

Mineral dissolution may be controlled by either mineral–melt interface reaction or mass transport of the mineral–constituent components in the melts.<sup>2,14</sup> Diffusive mineral

dissolution experiments provide diffusivity data for the equilibrium-determining component, the interface melt composition, and a way to model diffusive and convective dissolution rates as a function of temperature and pressure.<sup>2,15–19</sup> SiO<sub>2</sub> diffusion in silicate melts plays a major role not only in quartz dissolution/growth but also in a wide spectrum of igneous processes, including magma assimilation, double-diffusive convection, and fluid transport of SiO<sub>2</sub>. Extensive studies on SiO<sub>2</sub> diffusion, both self-diffusion and effective binary diffusion,<sup>20</sup> have been carried out.<sup>1,2,15–19,21–33</sup> However, prediction of SiO<sub>2</sub> diffusivity as a function of melt composition and temperature is still not available (see review in ref 34).

To quantify the kinetics of quartz dissolution in silicate melts and SiO<sub>2</sub> diffusion during the dissolution process, we have conducted two series of quartz dissolution experiments, one in rhyolitic melt and the other in basaltic melt. The results will be combined with literature data to quantify the saturation conditions of quartz and SiO<sub>2</sub> diffusivity as a function of melt composition during quartz dissolution in rhyolitic, andesitic, and basaltic melts. In addition, quartz dissolution and growth rates in silicate melts will be discussed.

## 2. SAMPLES, EXPERIMENTS, AND ANALYSES

Gem-quality quartz crystals and natural and synthetic rhyolitic and basaltic glasses were used in the dissolution experiments conducted in a 0.5 in. piston-cylinder apparatus. The quartz crystals are essentially pure SiO<sub>2</sub> (Table 1). For quartz dissolution experiments, two different starting glasses (melts) were used: one is rhyolite from Newberry Crater (NCO in Table 1) and the other is a midocean ridge basalt from Juan de Fuca Ridge (JDF in Table 1). Both are nominally dry. The basalt is a MORB glass sample from the Juan de Fuca Ridge,<sup>35,36</sup> the same as that used in refs 15, 16, and 18 for olivine, diopside, and anorthite dissolution experiments. Literature data on quartz dissolution in an andesitic melt<sup>2</sup> and on cassiterite dissolution in rhyolitic melts<sup>17</sup> will also be used to examine SiO<sub>2</sub> diffusivity in andesitic melt and hydrous rhyolitic melts; their compositions are also listed in Table 1. The rhyolitic glasses of NCO, CIT, KS, and H6a contain 0.1, 0.1, 0.9, and 5.9 wt % H<sub>2</sub>O, respectively. NCO, CIT, and KS are natural rhyolites,<sup>37</sup> and H6a is a synthetic rhyolitic glass prepared in the study of Hui et al.<sup>38</sup>

Starting quartz crystals were sectioned into small round disks with top and bottom surfaces perpendicular to its *c*-axis. The disks are ~1 mm in thickness and 2.4–2.6 mm in diameter. Rhyolitic and basaltic glasses were ground into cylinders about ~2 mm in length and 2.4–2.6 mm in diameter. The polished faces of a crystal disk and a glass cylinder were stacked together and then packed into a graphite capsule. The polished surfaces at the contact help maintain better contact between glass (to be melted) and quartz crystal when compressed (e.g., minimizing the chance of bubbles at the interface). To ensure gravitational stability in the melt, the quartz disk was placed on top of the glass cylinder. The capsule design is similar to that used in previous plagioclase dissolution experiments.<sup>18</sup> All experiments were run in a piston-cylinder apparatus with 0.5 in. piston using MgO sleeves and BaCO<sub>3</sub> cell.

According to the SiO<sub>2</sub> phase diagram (Figure 6 in ref 39),  $\beta$ -quartz is stable between ~750 and 1700 °C at 0.5 GPa. Hence, our quartz dissolution experiments at 1300–1600 °C and 0.5 GPa were carried out in  $\beta$ -quartz stability field. At all

experimental temperature and pressure conditions, the corresponding glass samples were molten. Experimental durations were chosen to be long enough to obtain measurable concentration profiles but short enough to satisfy the semi-infinite boundary condition. Time-series experiments were conducted to rule out convection in the melt and to check the reproducibility of our experiments. A summary of all the experimental conditions is listed in Table 2. During each experimental run, the temperature and pressure were maintained automatically using a programmed temperature controller and a pressure controller. The fluctuation in the experimental temperature maintained by the controller is  $\pm 1$  °C and that in pressure is  $\pm 4$  MPa. The experimental procedure has been detailed in ref 18. The temperature gradient is zero at the center of the graphite furnace (roughly the position of the interface) at, about 9 K/mm at 1 mm off the center and 18 K/mm at 2 mm off the center.<sup>38</sup> Reported temperatures and pressures in Table 2 have been corrected using the temperature calibration of ref 38 and pressure calibration of ref 40.

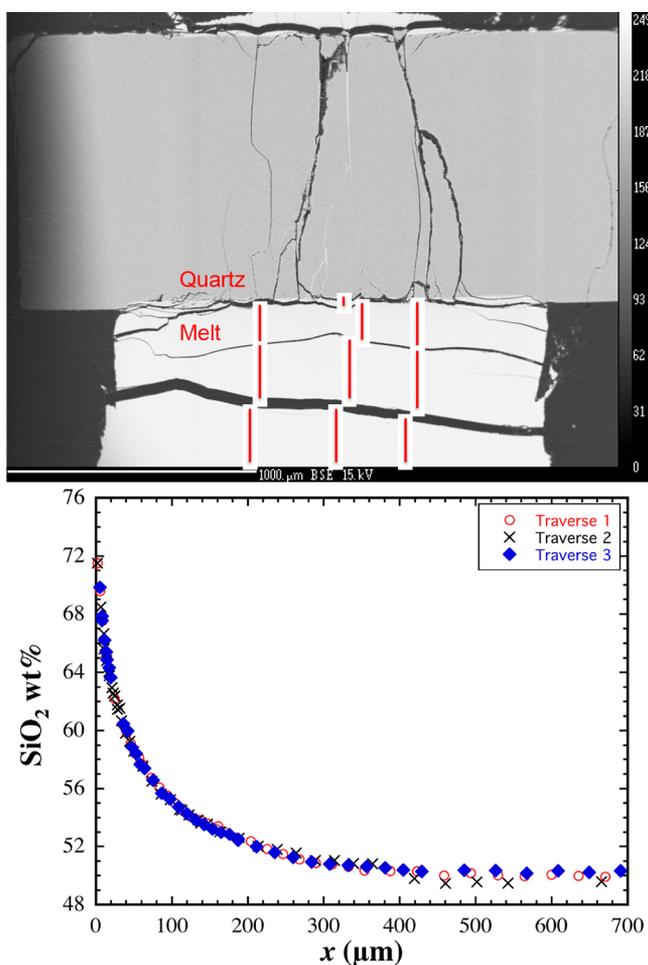
After quench of each experiment, the sample assembly was mounted in epoxy and then ground and polished along the central axis of symmetry until the exposed area of the sample is near maximum. The polished sample was further cleaned using ultrasonic cleaner, dried in a vacuum oven, and then carbon-coated before the electron microprobe analyzer (EMPA) analysis using a Cameca SX100 electron microprobe at the University of Michigan.

The EMPA analysis condition was 15 kV and 5 nA focused beam. Counting times are 60 s for Si, 40 s for Al, 30 s for Fe, Ca, and K, 20 s for Mg and Ti, and 10 s for Mn. Na was counted as one of the first elements on a spectrometer for six 5 s periods, and Na concentration is calculated by extrapolation to zero time. Three to four concentration traverses in the glass next to quartz were measured for each experimental sample from the mineral–glass interface to the far-field melt. The effect of quench cracks in the glass on distances of measured points to interface was corrected by cross-matching different traverses across the cracks. The electron microprobe data of all experimental samples is provided in the Supporting Information.

To augment our data, a quartz dissolution experiment in andesitic melt<sup>2</sup> is reexamined in this study. Furthermore, some cassiterite dissolution experiments in ref 17 also generated significant SiO<sub>2</sub> concentration profiles and are examined in this work for SiO<sub>2</sub> diffusivity. The cassiterite dissolution experiments were carried out in rhyolitic melt containing 0.1–5.9 wt % H<sub>2</sub>O and at 750–1100 °C and 0.5 GPa. The goal of their experiments was to investigate tin diffusion and cassiterite dissolution kinetics. The resulting profiles in five experiments (CassDis1, 3, 8, 10, and 11) display significant variations in SiO<sub>2</sub> concentration, providing constraints on the effect of H<sub>2</sub>O on SiO<sub>2</sub> diffusion and are used in this study. Table 2 also includes a summary of these experiments.

## 3. EXPERIMENTAL RESULTS

Run conditions of all experiments are summarized in Table 2. We conducted 12 experiments on quartz dissolution in NCO rhyolitic melt and 7 experiments on quartz dissolution in JDF basaltic melt. Figure 1 shows a back scatter electron (BSE) image of a typical sample with EMPA measurement traverses indicated as red line segments. By comparing all measured traverses within the same experiment, the corresponding



**Figure 1.** (Top) back-scattered electron image of the experiment sample QzDisBa101 with three measured traverses (red lines) marked on it. (Bottom)  $\text{SiO}_2$  concentration profiles for the same sample.

concentration profiles are consistent within error (Figure 1), supporting the absence of convection during the experiments. Consistency among time-series experiments further supports absence of convection.

In quartz dissolution experiments,  $\text{SiO}_2$  concentration decreases from the interface melts toward the far-field melts (Figure 1). Figures 2 and 3 show examples of oxide concentration profiles during quartz dissolution in molten rhyolite and basalt, respectively. As quartz dissolves into a silicate melt, the  $\text{SiO}_2$  concentration increases toward the quartz–melt interface, and the concentrations of other components (e.g.,  $\text{TiO}_2$ ,  $\text{Al}_2\text{O}_3$ ,  $\text{FeO}$ ,  $\text{MgO}$ ,  $\text{CaO}$ ,  $\text{Na}_2\text{O}$ , and  $\text{K}_2\text{O}$ ) are expected to be diluted and to decrease toward the interface. For quartz dissolution into rhyolitic melts (Figure 2), the profiles of these components exhibit the “expected” pattern. During quartz dissolution in basaltic melt (Figure 3), the profiles of  $\text{SiO}_2$ ,  $\text{TiO}_2$ ,  $\text{Al}_2\text{O}_3$ ,  $\text{FeO}$ ,  $\text{MgO}$ , and  $\text{CaO}$  also exhibit the “expected” behavior. However,  $\text{Na}_2\text{O}$  and  $\text{K}_2\text{O}$  profiles show complex uphill diffusion patterns in the basaltic melt toward quartz–melt interface where the  $\text{SiO}_2$  concentration is high. The uphill diffusion of  $\text{Na}_2\text{O}$  and  $\text{K}_2\text{O}$  toward the quartz–melt interface is expected to lead to a minimum in the profile (e.g., Figure 9C in ref 41) to satisfy mass balance. However, the minimum is not obvious in  $\text{Na}_2\text{O}$  and  $\text{K}_2\text{O}$  profiles but  $\text{Na}_2\text{O}$  profile shows a maximum. For

components with normal diffusion pattern, the concentration profiles usually cannot be fit well by assuming a constant diffusivity (solid red curves in Figures 2 and 3) especially in basaltic melts. Furthermore, the relative lengths of the concentration profiles for the various oxides do not match previously observed differences in the tracer diffusivities:<sup>34</sup> for example, in Figure 2 the  $\text{K}_2\text{O}$  profile is the shortest and  $\text{Na}_2\text{O}$  profile is about the same length as the  $\text{SiO}_2$  profile, even though Na and K tracer diffusivities are much higher than  $\text{SiO}_2$  diffusivity.<sup>34</sup> These features demonstrate that the diffusion of other components in the systems is largely controlled by  $\text{SiO}_2$  diffusion as well as the effects of multicomponent diffusion.<sup>1,33,41</sup> In cassiterite dissolution experiments,<sup>17</sup>  $\text{SiO}_2$  concentration decreases toward the cassiterite–melt interface as the dissolution of cassiterite dilutes all major components in the melt.

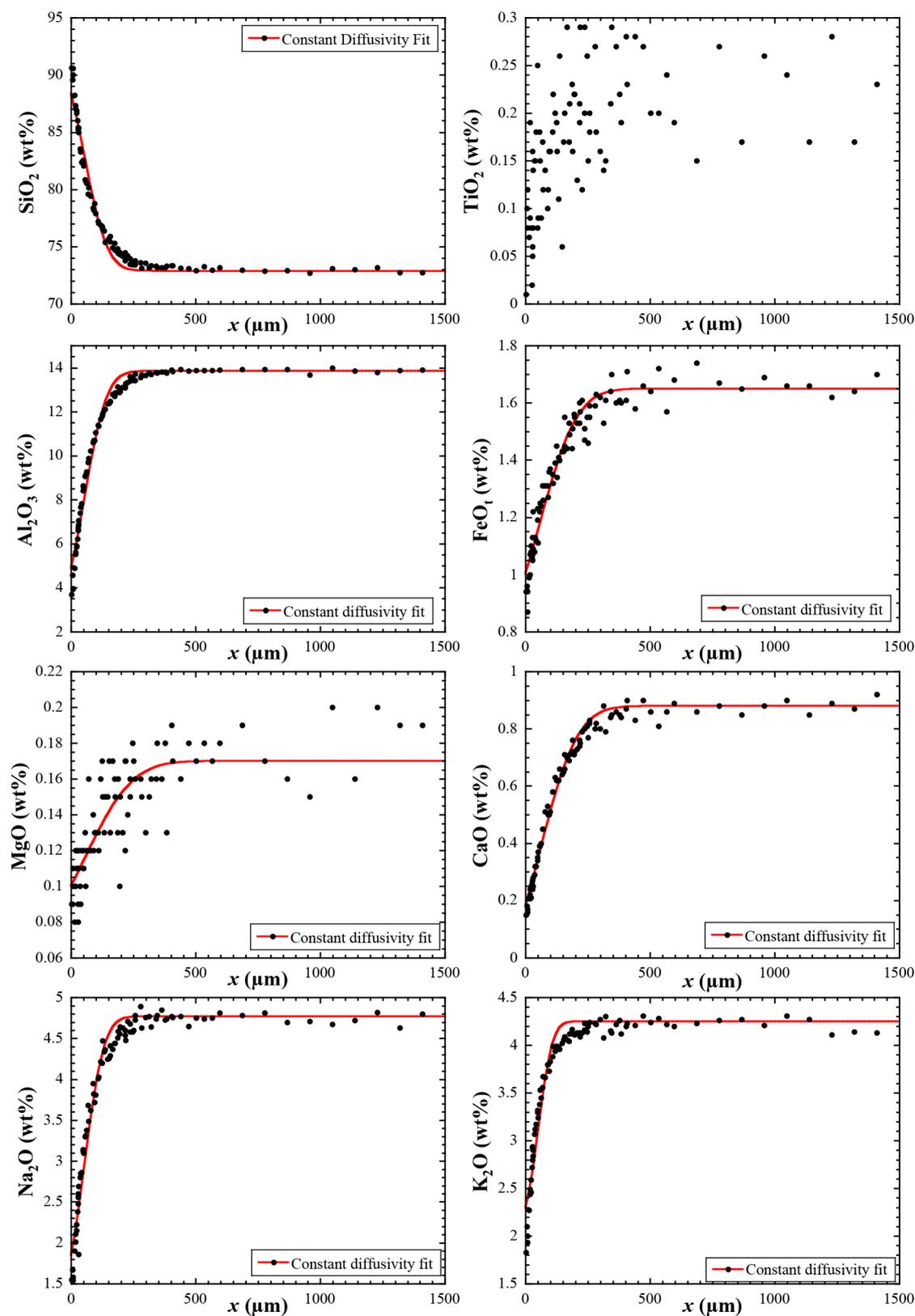
A summary of interface–melt compositions of all the experiments can be found in Table 3. Figure 4 shows temporal variations of  $\text{SiO}_2$  concentration at the interface melt and melt growth distances at different temperature conditions for quartz dissolution. The interface  $\text{SiO}_2$  concentrations were obtained by fitting the  $\text{SiO}_2$  concentration profiles and extrapolating the profile to the interface. The fitting procedure will be discussed later. For quartz dissolution in rhyolite, the  $\text{SiO}_2$  concentration in the interface melt is roughly the same within error at a given experimental temperature (1300, 1400, and 1500 °C) regardless of the experimental duration. For quartz dissolution in basalt experiments, the interface melt concentration of  $\text{SiO}_2$  is roughly constant at 1300 °C. There is relatively large variation for the two experiments at 1400 °C (Figure 4b), likely due to both experimental uncertainty (e.g., in the accuracy of the experimental temperature) and uncertainty in extrapolating  $\text{SiO}_2$  concentrations to estimate its concentration at the interface melts. Note that the interface melt composition cannot be measured directly even if new instrumentation allows higher spatial resolution because the interface melt were changed were changed by quartz dissolution during quench.<sup>2</sup>

The melt growth distance ( $L_{\text{melt}}$ ) and mineral dissolution distance ( $L_{\text{mineral}}$ ) were calculated based on mass balance (eq 16 in ref 2), with  $L_{\text{mineral}} = L_{\text{melt}}(\rho_{\text{melt}}/\rho_{\text{mineral}})$ . Because electron microprobe distances are measured at room temperature and pressure, glass density rather than melt density is used. Because the dissolving mineral during the experiments is  $\beta$ -quartz (not  $\alpha$ -quartz), the estimated density of  $\beta$ -quartz, about 2.54 g/cm<sup>3</sup> at room temperature and pressure<sup>39,42</sup> is used. That is, the reported crystal dissolution distance is for  $\beta$ -quartz dissolution. For cassiterite, the density is about 6.99 g/cm<sup>3</sup>. Figure 4c,d shows there are good linear relations ( $L = b\sqrt{t}$ ) between the melt growth distance ( $L$ ) and the square root of the experimental duration ( $\sqrt{t}$ ).

On the basis of both the constancy of the  $\text{SiO}_2$  concentration in the interface melt at a given temperature, and the proportionality between the melt growth distance and the square root of the experimental duration, we infer that quartz dissolution in both rhyolitic and basaltic melts is controlled by diffusion of  $\text{SiO}_2$ , implying rapid interface reaction.

#### 4. DISCUSSION

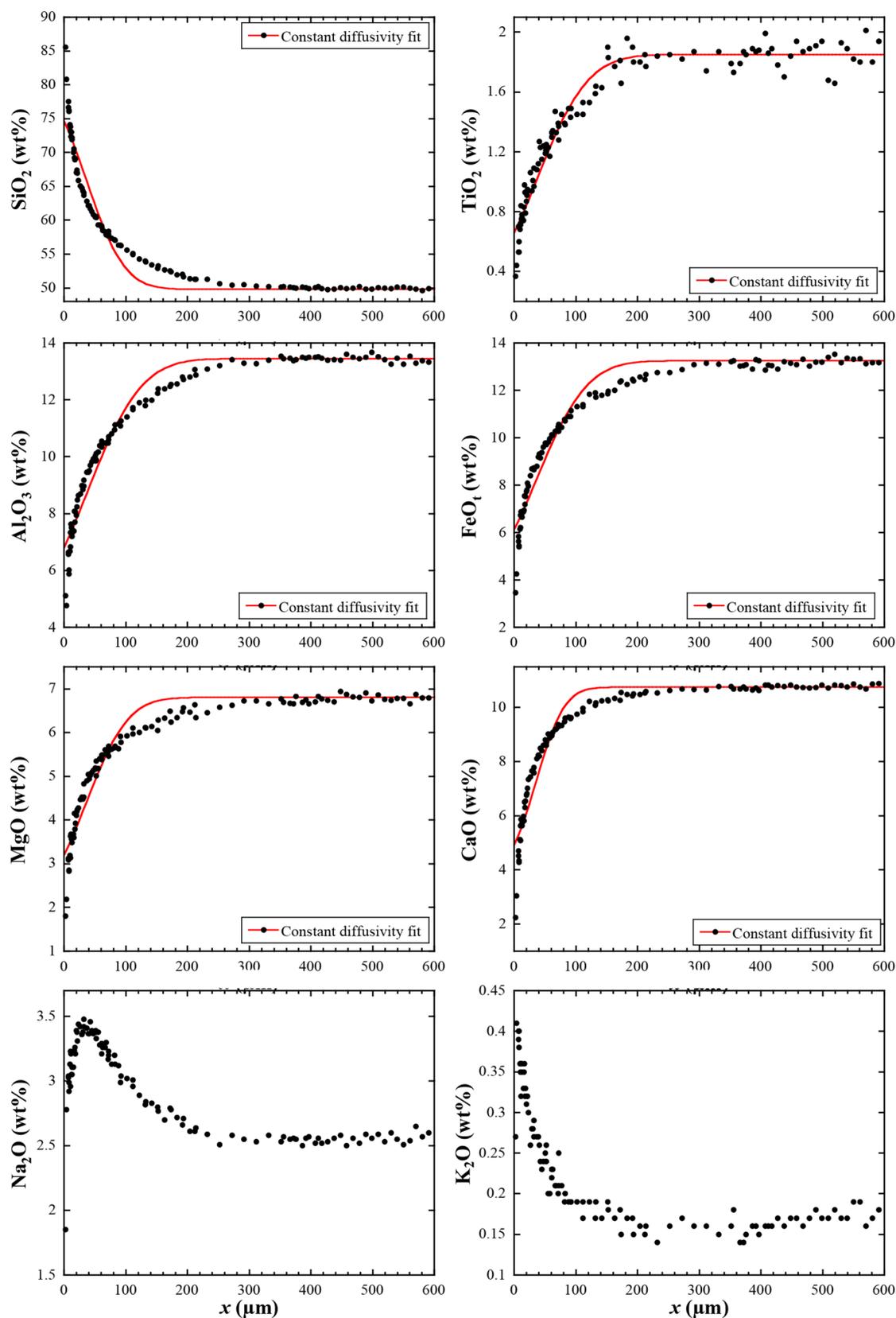
Diffusion in the melt during quartz dissolution is fundamentally a multicomponent diffusion problem. Major advancement



**Figure 2.** Oxide concentration profiles for a quartz dissolution experiment in a rhyolitic melt (QzDisRh104, 1501 °C, 0.5 GPa, 86432 s).

has been made in treating multicomponent diffusion in natural melts<sup>43–46</sup> but data in this study are not enough to apply a rigorous multicomponent diffusion treatment. One reason is that the experiments in our study do not cover enough

compositional directions. For example, refs 45–47 showed that experiments covering at least seven mutually perpendicular compositional directions (preferably more experiments) are needed for an eight-component melt system. Second, as it



**Figure 3.** Oxide concentration profiles for a quartz dissolution experiment in a basaltic melt (QzDisBa107, 1576 °C, 0.5 GPa, 322 s).

will be clear soon, the diffusion coefficients in the melt vary across a diffusion profile during quartz dissolution due to large variation in  $\text{SiO}_2$  concentration, meaning that the diffusion

matrix is not a constant matrix, further complicating the multicomponent diffusion treatment.

Table 3. Extrapolated Interface–Melt Compositions of All Experiments<sup>a,b</sup>

sample#	SiO <sub>2</sub>	SnO <sub>t</sub>	TiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	FeO <sub>t</sub>	MnO	MgO	CaO	Na <sub>2</sub> O	K <sub>2</sub> O	total
QzDisRh103	82.96		0.13	8.8	1.3	0.03	0.07	0.28	3.5	3.1	100.17
QzDisRh111	85.61		0.10	6.4	1.1	0.04	0.09	0.21	3.0	2.9	99.45
QzDisRh112	84.22		0.12	7.8	1.2	0.05	0.09	0.28	2.6	2.9	99.26
QzDisRh115	83.86		0.13	7.8	1.3	0.02	0.10	0.30	3.1	3.0	99.61
QzDisRh201	84.92		0.10	7.5	1.5	0.05	0.11	0.30	2.2	3.2	99.88
QzDisRh203	84.58		0.13	7.3	1.3	0.04	0.12	0.26	2.3	3.0	99.03
QzDisRh105	90.14		0.10	3.9	0.9	0.02	0.07	0.14	1.3	1.8	98.37
QzDisRh113	88.02		0.14	5.8	1.2	0.05	0.09	0.20	2.3	2.4	100.20
QzDisRh114	89.26		0.05	4.6	1.1	0.05	0.08	0.18	1.5	2.0	98.82
QzDisRh102	90.92		0.15	4.8	1.1	0.04	0.07	0.14	1.8	2.3	101.32
QzDisRh104	91.40		0.06	3.4	0.9		0.08	0.13	1.4	1.7	99.07
QzDisRh106	95.38		0.06	0.9	0.8		0.06	0.08	0.8	0.8	98.88
Andesite-234	82.55		0.44	8.0	2.0	0.05	1.0	1.7	1.9	2.45	100.09
QzDisBa101	73.54		0.82	8.1	6.1	0.09	3.10	5.00	3.6	0.5	100.81
QzDisBa102	71.52		0.78	8.1	6.3	0.09	3.15	4.90	4.0	0.5	99.29
QzDisBa103	71.87		0.73	8.5	6.6	0.10	3.30	5.50	3.7	0.4	100.70
QzDisBa110	80.29		0.55	5.9	4.0	0.03	2.20	3.20	2.9	0.5	99.52
QzDisBa111	75.99		0.72	7.1	5.6	0.06	2.80	4.20	3.1	0.5	100.02
QzDisBa104	90.02		0.26	5.1	2.2		0.50	0.00	0.9	0.5	99.48
QzDisBa107	92.66		0.32	3.5	1.4		0.50	0.00	1.1	0.5	99.98
CassDis8	66.39	5.00	0.25	12.2	2.0		0.28	1.57	3.4	3.8	94.89
CassDis10	70.71	5.37	0.06	11.1	1.3		0.03	1.30	3.7	4.2	97.77
CassDis3	68.62	9.56	0.06	11.2	1.3		0.03	1.00	3.3	4.0	99.07
CassDis1	64.15	12.83	0.06	10.8	1.4		0.03	1.10	3.3	3.8	97.47
CassDis11	66.00	12.49	0.05	10.3	1.3		0.02	1.20	3.5	3.5	98.36

<sup>a</sup>FeO<sub>t</sub> stands for total iron oxide expressed as FeO, and similar for SnO<sub>t</sub>. Interface–melt concentrations were estimated by visual fits of the concentration profiles to the interface except for SiO<sub>2</sub> whose values were from fitting SiO<sub>2</sub> concentration profiles using eq 4, and for SnO<sub>t</sub> for cassiterite dissolution experiments. <sup>b</sup>Cassiterite dissolution experiments with no discernible or low-quality SiO<sub>2</sub> concentration profiles are not included.

The main purpose of this work is to understand the kinetics of quartz dissolution and quantify diffusive dissolution rates, which are largely controlled by SiO<sub>2</sub> diffusion. The concentration profiles of SiO<sub>2</sub> in the melt during quartz dissolution are smooth and monotonic, which lend themselves to the effective binary diffusion approach.<sup>20</sup> This is consistent with the conclusion of ref 2 that the diffusion of the principal equilibrium-determining component (SiO<sub>2</sub> in the case of quartz dissolution) can be roughly treated as effective binary. The diffusion of other components is of interest but less important for understanding quartz dissolution kinetics, and hence is not treated in this work. Nonetheless, the complete electron microprobe data of all oxides are provided in Supporting Information, which may be used for multi-component diffusion treatment when more data are available or for other purposes. Below, we will first discuss the fitting of SiO<sub>2</sub> concentration profiles during quartz dissolution to obtain the effective binary diffusivity (EBD) of SiO<sub>2</sub>. We show that EBD of SiO<sub>2</sub> depends strongly on melt composition, in agreement with previous authors.<sup>1,21,23,28,33</sup> We fit SiO<sub>2</sub> concentration profile during quartz dissolution and make an effort to quantify how SiO<sub>2</sub> diffusivity depends on melt composition. The fitting of SiO<sub>2</sub> diffusion profile also provides the SiO<sub>2</sub> concentration at the quartz–melt interface, which approximates the SiO<sub>2</sub> concentration at quartz saturation (SCQS). The variation of SCQS will be examined, and the results will be useful for constraining thermodynamic models for silicate melts (such as the MELTS model<sup>9–13</sup>). We then develop a method to quantify quartz dissolution rate considering composition-dependent SiO<sub>2</sub> diffusivity, which

has general applicability to dissolution or growth of other minerals when the EBD of the equilibrium-determining component varies along the diffusion profile (e.g., cassiterite dissolution<sup>17</sup>).

**4.1. Fitting SiO<sub>2</sub> Concentration Profiles from Quartz Dissolution Experiments.** For diffusion-controlled mineral dissolution, if the effective binary diffusivity of a given component is constant across the whole concentration profile, the concentration profile can be fit by the following equation<sup>2</sup>

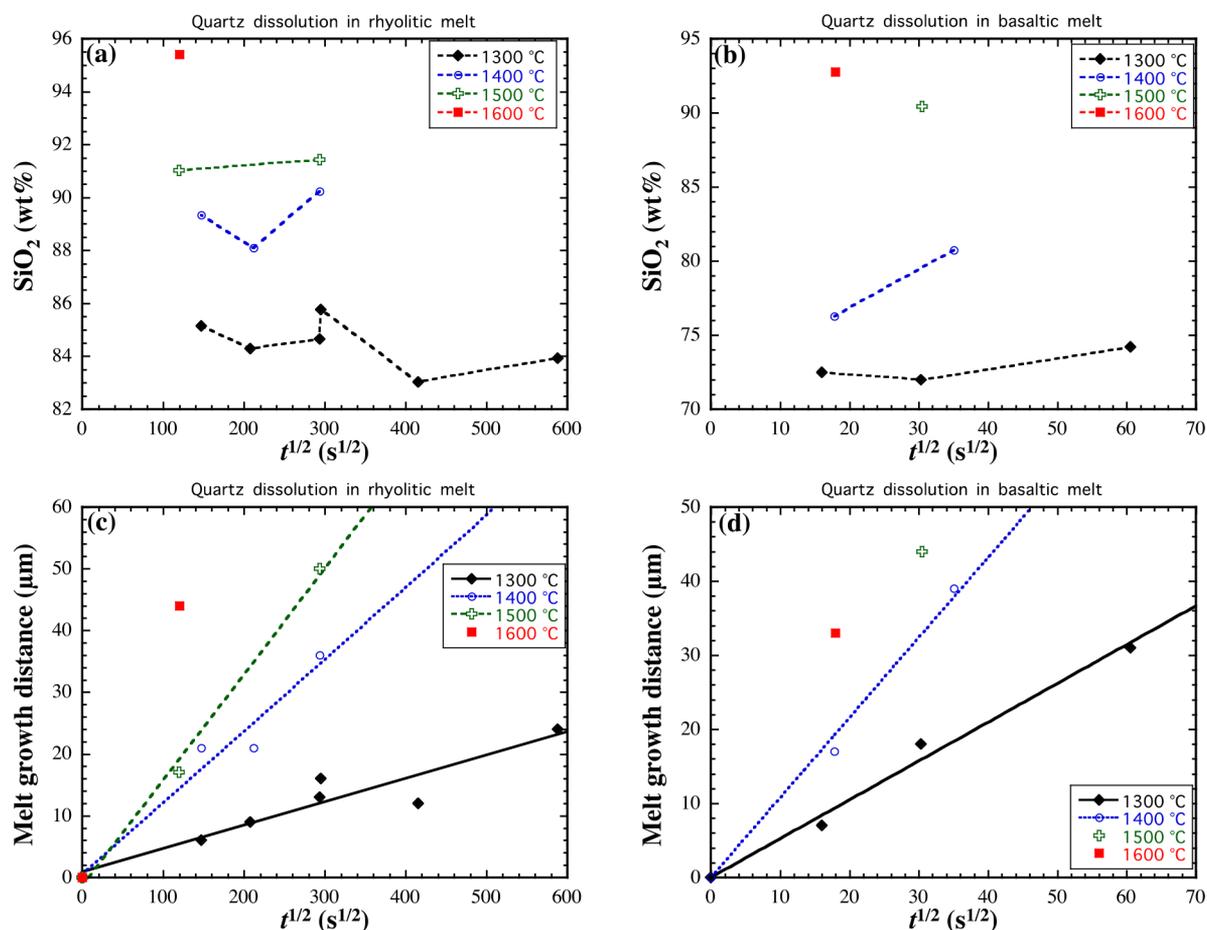
$$C = C_{\infty} + (C_0 - C_{\infty}) \frac{\operatorname{erfc}\left(\frac{x}{\sqrt{4Dt}} - \alpha\right)}{\operatorname{erfc}(-\alpha)} \quad (1)$$

in which  $\alpha$  is solved from the following

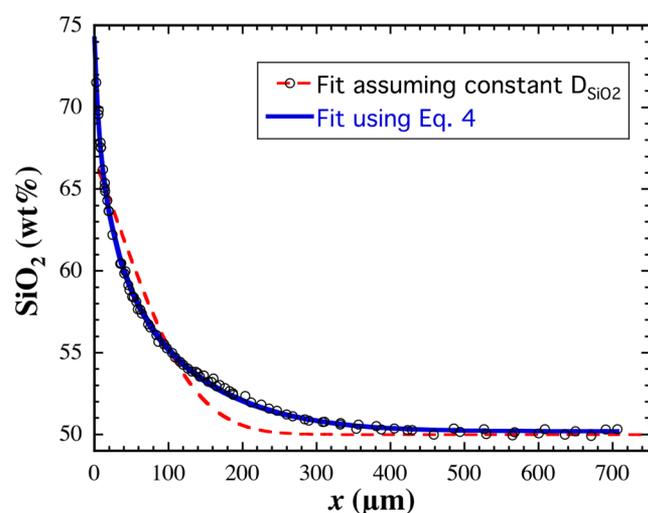
$$\sqrt{\pi} \alpha e^{\alpha^2} \operatorname{erfc}(-\alpha) = \frac{C_0 - C_{\infty}}{C_c - C_0} \quad (2)$$

where  $C_{\infty}$ ,  $C_0$ ,  $C_c$ , and  $C$  are the concentrations of the diffusing component in the far-field melt, interface melt, the crystal, and the melt at distance  $x$  away from the mineral–melt interface,  $D$  is the EBD of the diffusing component, and  $t$  is the experimental duration.

However, a comparison (Figures 2, 3, and 5) between the SiO<sub>2</sub> concentration data and the constant-diffusivity model (eq 1) shows obvious misfit (more so for quartz dissolution in basalt), indicating that there is significant variation in the SiO<sub>2</sub> EBD across the concentration profile. Furthermore, the misfit is systematic with data forming steeper trend at higher SiO<sub>2</sub> and shallower trend at lower SiO<sub>2</sub> compared to the constant-diffusivity curve. This indicates that  $D$  decreases with



**Figure 4.** (a,b) SiO<sub>2</sub> concentrations in the melt at quartz–melt interface versus the experimental duration for quartz dissolution experiments in rhyolite and basalt respectively. (c,d) Variation of melt growth distance with time for the two sets of quartz dissolution experiments. Melt growth distances were calculated using mass balance.



**Figure 5.** Typical SiO<sub>2</sub> concentration profile (black open circles, sample QzDisBa101) and fitting results using constant diffusivity of eq 1 (red dash curve) and composition-dependent diffusivity of eq 4 (blue solid curve).

increasing SiO<sub>2</sub>, as previously demonstrated.<sup>1,21,23,28,33</sup> Therefore, it is necessary to quantify how the EBD of SiO<sub>2</sub> varies along a profile. Below, we will use both the Boltzmann analysis and the fit of concentration profiles using a functional

dependence of  $D(C)$  to determine the compositional dependence of SiO<sub>2</sub> EBD.

**4.1.1. Boltzmann Analysis of Concentration Profiles from Quartz Dissolution Experiments.** In diffusion-couple experiments, Boltzmann–Matano analysis<sup>48</sup> or the Sauer-Freie analysis<sup>49</sup> has been applied to investigate the compositional dependence of diffusivity.<sup>1,21,23,28,33,50</sup> The quality of such an approach largely depends on the precision of the original data of the concentration profiles. It has been shown<sup>1,21,23,28,33</sup> that  $\ln D_{\text{SiO}_2}$  depends roughly linearly on the SiO<sub>2</sub> concentration. Below, we use a similar Boltzmann approach, and derive the following equation (Appendix A) to calculate the diffusivity at each point (also meaning each concentration) using concentration profiles generated by mineral dissolution experiments

$$D = \frac{\frac{C_\infty - C_x}{C_c - C_\infty} \int_{C_0}^{C_\infty} x \, dC + \int_{C_x}^{C_\infty} x \, dC}{2t \left. \frac{\partial C}{\partial x} \right|_x}, \quad t > 0 \quad (3)$$

where  $C_x$  is SiO<sub>2</sub> concentrations in the melt at distance  $x$  away from the mineral–melt interface. Note here that the interface position is known and fixed for the case of mineral dissolution experiments, rather than the adjustable Matano interface<sup>48</sup> for the case of diffusion couple. In addition, diffusion must not

Table 4. Summary of Fitting Results for Quartz Dissolution Experiments Based on Equation 4<sup>a</sup>

exp#	$\beta$	$B$ ( $10^{-6}$ m/s <sup>1/2</sup> )	$C_0$ (wt %)	$\ln D_{X=1}$ (m <sup>2</sup> /s)	$\ln D_{\text{farfield}}$ (m <sup>2</sup> /s)	$\ln D_{\text{interface}}$ (m <sup>2</sup> /s)	$\ln D_{\text{eff}}$ (m <sup>2</sup> /s)
QzDisRh103	23.92(19)	0.0155(3)	82.96(19)	-36.60(23)	-32.42(21)	-33.60	-33.19
QzDisRh111	23.92(19)	0.0235(4)	85.61(20)	-36.04(23)	-31.92(21)	-33.50	-32.89
QzDisRh112	23.92(19)	0.0216(3)	84.22(11)	-35.69(23)	-31.95(21)	-33.25	-32.78
QzDisRh115	23.92(19)	0.0211(3)	83.86(10)	-36.12(23)	-31.81(21)	-33.34	-32.76
QzDisRh201	23.92(19)	0.0221(4)	84.92(32)	-35.76(23)	-32.03(21)	-33.45	-32.90
QzDisRh203	23.92(19)	0.0209(2)	84.58(10)	-35.96(23)	-31.98(21)	-33.53	-32.93
QzDisRh105	22.75(15)	0.0595(5)	90.14(9)	-34.28(19)	-30.56(17)	-32.71	-31.89
QzDisRh113	22.75(15)	0.0464(8)	88.02(19)	-34.85(19)	-30.82(17)	-32.70	-32.00
QzDisRh114	22.75(15)	0.0695(7)	89.26(14)	-33.88(19)	-30.22(17)	-32.17	-31.44
QzDisRh102	21.55(12)	0.0717(9)	90.92(31)	-34.02(15)	-30.18(14)	-32.59	-31.65
QzDisRh104	21.55(12)	0.0809(8)	91.40(11)	-33.70(16)	-30.10(14)	-32.34	-31.49
QzDisRh106	19.53(14)	0.1723(13)	95.83(17)	-32.38(17)	-29.13(16)	-31.64	-30.73
QzDisBa101	23.92(19)	0.2376(28)	73.54(23)	-35.52(21)	-26.28(16)	-30.12	-28.28
QzDisBa102	23.92(19)	0.2794(23)	71.52(23)	-35.11(21)	-25.87(16)	-29.45	-27.75
QzDisBa103	23.92(19)	0.2729(25)	71.87(25)	-35.21(21)	-25.98(16)	-29.36	-27.82
QzDisBa110	22.75(15)	0.4992(51)	80.29(27)	-33.73(17)	-24.96(13)	-30.01	-27.51
QzDisBa111	22.75(15)	0.4729(47)	75.99(22)	-33.79(17)	-24.99(13)	-29.15	-27.18
QzDisBa104	21.55(12)	0.7018(54)	90.02(42)	-32.72(14)	-24.45(11)	-30.92	-27.77
QzDisBa107	19.53(14)	0.9570(82)	92.66(48)	-31.72(16)	-24.17(13)	-30.31	-27.44

<sup>a</sup>The values in the parentheses indicate  $1\sigma$  errors on the last digit based on the fitting, but the true error is best gauged from experiments at the same temperature (same  $\beta$  value).

have reached the far-field so that the diffusion medium is semi-infinite.

To implement the above approach, a MatLab program was written in which a profile is first smoothed using a moving average filter with a span of 10 points or more. The integral and slope in eq 3 are calculated using the smoothed profile. Then  $D$  at a given position (or given concentration) is calculated using eq 3. Near the far-field, concentration slopes in the denominator of eq 3 approach zero and cannot be evaluated accurately. Near the interface, the concentration profile is steep and the slopes are not well constrained. Hence, the calculated  $D$  values at both ends of the concentration profile are usually less accurate and show large fluctuations. To examine the compositional dependence of  $D$ , we plot  $\ln D$  versus  $\text{SiO}_2$  concentration as in previous studies.<sup>1,21,23,28,33</sup> Only the smooth part of the trend in the middle concentration range (e.g., for exp. QzDisBa110,  $C_\infty = 50.0$  wt %,  $C_0 = 80.3$  wt %, but only  $D$  values from 52 to 74 wt %  $\text{SiO}_2$ ) is chosen. The results are plotted in Figure 6.

We first examine how  $\ln D_{\text{SiO}_2}$  depends on  $\text{SiO}_2$  concentration (left-hand side in Figure 6). We note that  $\ln D_{\text{SiO}_2}$  is roughly linear to  $\text{SiO}_2$  concentration (meaning that  $D_{\text{SiO}_2}$  is roughly an exponential function of  $\text{SiO}_2$  concentration) in a given starting melt and at a given temperature, consistent with previous results.<sup>1,21,23,28,33</sup> However, there is a significant offset among the trends (i.e., different intercepts) at the same temperature when comparing the trends across different initial melts (rhyolite, andesite, and basalt) with  $\text{SiO}_2$  content as the compositional parameter, indicating that other compositional parameters also influence  $\text{SiO}_2$  diffusion.

We next examine  $\ln D$  versus  $X_{\text{Si+Al}}$  plot (right-hand side of Figure 6), where  $X_{\text{Si+Al}}$  is the sum of cation mole fractions of Si and Al (for the benefit of readers, the Supporting Information contains examples for the calculation of  $X_{\text{Si+Al}}$  for basalt far-field composition). The data show that using  $X_{\text{Si+Al}}$  as the compositional parameter reduces the disparity among the different melts. A single trend approximately

captures the  $\text{SiO}_2$  diffusion behavior across rhyolitic, andesitic, and basaltic melts considering uncertainties as gauged by multiple experiments of quartz dissolution in rhyolitic melts. These observations indicate that  $X_{\text{Si+Al}}$  (simplified as  $X$  hereafter) is a better parameter to characterize the compositional effect on  $\text{SiO}_2$  EBD than using  $\text{SiO}_2$  alone. That  $D_{\text{SiO}_2}$  depends on the cation mole fraction of both Si and Al is not surprising because both Si and Al are network formers controlling melt structure and viscosity. In plagioclase dissolution experiments, ref 18 reached a similar conclusion that Si and Al together control the diffusivities of  $\text{Al}_2\text{O}_3$ . Reference 19 also showed that Zr diffusivity depends on Si+Al rather than on  $\text{SiO}_2$  alone. Nonetheless, there are still minor offsets in the  $\ln D_{\text{SiO}_2}$  versus  $X$  linear trends for different melts; the maximum offset is about 1.1  $\ln D$  units at 1500 °C. The offsets are largely within data uncertainty, but they may also be due to the effect of other compositional parameters (e.g., alkali earth elements and alkali elements) on  $D_{\text{SiO}_2}$ , which cannot be resolved currently.

**4.1.2. Fitting  $\text{SiO}_2$  Profiles by Using Exponential Dependence of  $\text{SiO}_2$  Diffusivity.** The major advantage of using Boltzmann analysis to extract diffusivity as a function of composition is that no prescribed functional dependence is assumed and hence the relation is unbiased. However, this method is sensitive to the smoothness of a concentration profile. In addition, there is some arbitrariness in choosing which part of the  $D(C)$  relation is the reliable part. To further verify and quantify the relation between  $\text{SiO}_2$  EBD and melt composition, we fit the  $\text{SiO}_2$  concentration profiles assuming  $D$  depends on  $C$  or  $X$  exponentially based on insights from the preceding section and from the literature.<sup>1,21-28,33,34,50-52</sup> The following diffusivity relation is adopted

$$D = D_{\text{interface}} e^{-\beta(C-C_0)} = D_{\text{farfield}} e^{-\beta(C-C_\infty)} = D_{C=100} e^{\beta(100-C)} \quad (4a)$$

$$D = D_{\text{interface}} e^{-\beta(X-X_0)} = D_{\text{farfield}} e^{-\beta(X-X_\infty)} = D_{X=1} e^{\beta(1-X)} \quad (4b)$$

Table 5. Summary of  $\ln D_{\text{SiO}_2}$  from Fitting Results Based on Equation 4

exp#	$\beta$	$\ln D_{X=1}$ (m <sup>2</sup> /s)	$\ln D$ (m <sup>2</sup> /s)						
			$X = 0.65$	$X = 0.70$	$X = 0.75$	$X = 0.80$	$X = 0.85$	$X = 0.90$	$X = 0.95$
QzDisRh103	23.92(19)	-36.60(23)	-28.23	-29.43	-30.62	-31.82	-33.01	-34.21	-35.41
QzDisRh111	23.92(19)	-36.04(23)	-27.67	-28.87	-30.07	-31.26	-32.46	-33.65	-34.85
QzDisRh112	23.92(19)	-35.69(23)	-27.32	-28.52	-29.71	-30.91	-32.10	-33.30	-34.50
QzDisRh115	23.92(19)	-36.12(23)	-27.75	-28.94	-30.14	-31.34	-32.53	-33.73	-34.92
QzDisRh201	23.92(19)	-35.76(23)	-27.39	-28.59	-29.78	-30.98	-32.17	-33.37	-34.57
QzDisRh203	23.92(19)	-35.96(23)	-27.59	-28.79	-29.98	-31.18	-32.37	-33.57	-34.77
QzDisRh105	22.75(15)	-34.28(19)	-26.31	-27.45	-28.59	-29.73	-30.86	-32.00	-33.14
QzDisRh113	22.75(15)	-34.85(19)	-26.89	-28.03	-29.17	-30.30	-31.44	-32.58	-33.72
QzDisRh114	22.75(15)	-33.88(19)	-25.92	-27.06	-28.19	-29.33	-30.47	-31.61	-32.74
QzDisRh102	21.55(12)	-34.02(15)	-26.48	-27.56	-28.63	-29.71	-30.79	-31.87	-32.94
QzDisRh104	21.55(12)	-33.70(16)	-26.16	-27.23	-28.31	-29.39	-30.47	-31.54	-32.62
QzDisRh106	19.53(14)	-32.38(17)	-25.54	-26.52	-27.50	-28.47	-29.45	-30.43	-31.40
QzDisBa101	23.92(19)	-35.52(21)	-27.15	-28.35	-29.54	-30.74	-31.94	-33.13	-34.33
QzDisBa102	23.92(19)	-35.11(21)	-26.74	-27.94	-29.13	-30.33	-31.53	-32.72	-33.92
QzDisBa103	23.92(19)	-35.21(21)	-26.84	-28.03	-29.23	-30.42	-31.62	-32.81	-34.01
QzDisBa110	22.75(15)	-33.73(17)	-25.77	-26.91	-28.05	-29.18	-30.32	-31.46	-32.60
QzDisBa111	22.75(15)	-33.79(17)	-25.83	-26.96	-28.10	-29.24	-30.38	-31.51	-32.65
QzDisBa104	21.55(12)	-32.72(14)	-25.18	-26.25	-27.33	-28.41	-29.49	-30.56	-31.64
QzDisBa107	19.53(14)	-31.72(16)	-24.89	-25.86	-26.84	-27.82	-28.79	-29.77	-30.75

<sup>a</sup>SiO<sub>2</sub> diffusivities at specific melt compositions ( $X_{\text{Si+Al}} = 0.65, 0.7, 0.75, 0.8, 0.85, 0.9, 0.95$ ) are calculated based on eq 4. The values in the parentheses indicate  $1\sigma$  errors on the last digit based on the fitting.

where  $C$  is SiO<sub>2</sub> wt %,  $X = X_{\text{Si+Al}}$ ,  $D$ ,  $D_{\text{interface}}$ ,  $D_{\text{farfield}}$ ,  $D_C = 100$ , and  $D_{X=1}$  are the SiO<sub>2</sub> diffusivity in the melt of composition  $C$ , interface melt ( $C_0$ ), the far-field melt ( $C_\infty$ ), 100% SiO<sub>2</sub>, and when  $X = 1$ ;  $\beta'$  and  $\beta$  characterize the dependence of the diffusivity on SiO<sub>2</sub> (wt %) and  $X$ . Our approach here is similar to that in refs 17 and 33, though in ref 17 Sn and Si diffusivities near the interface are higher than that in the far-field melt, whereas during quartz dissolution Si diffusivity near the interface is lower than that in the far-field melt.

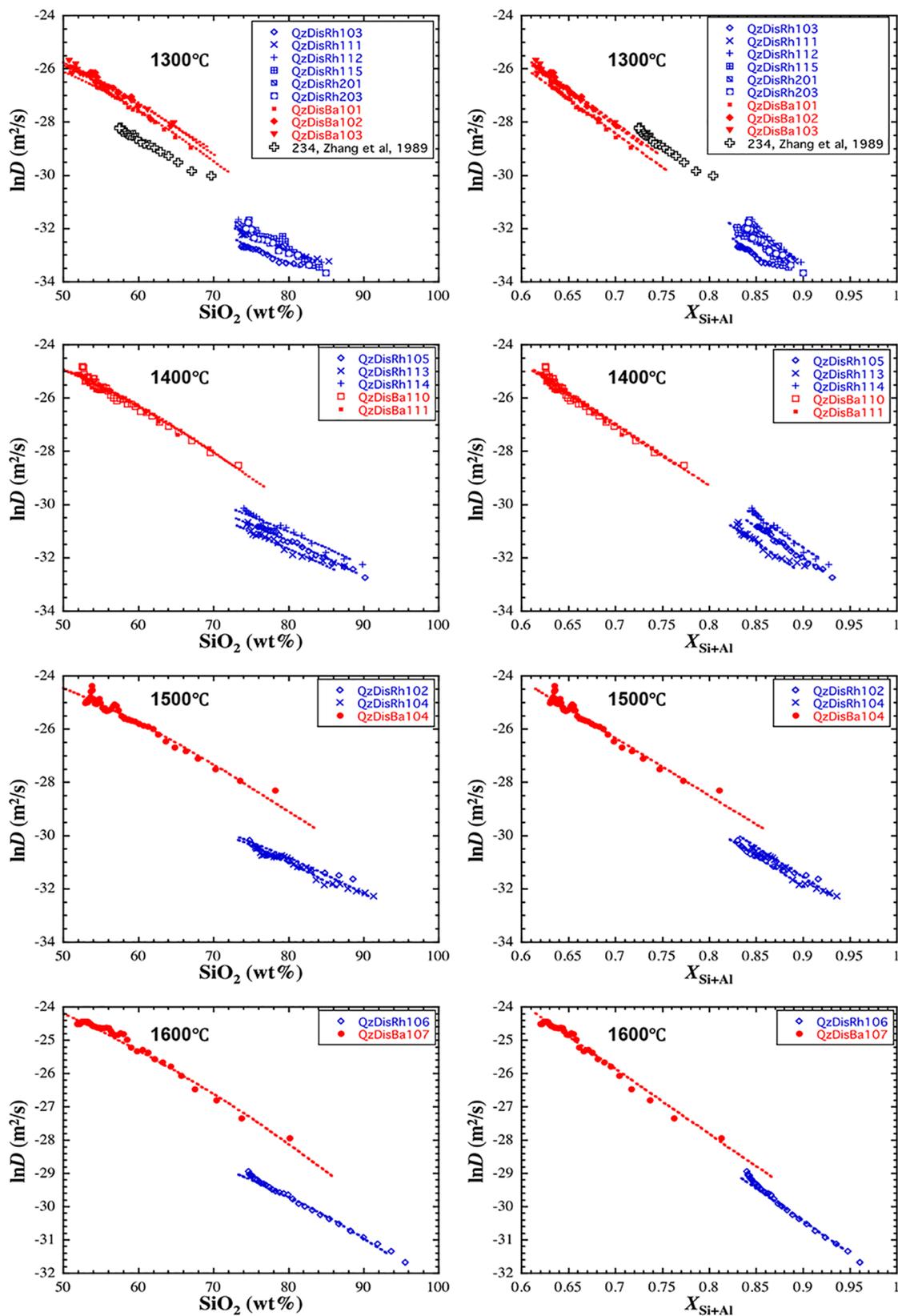
In the model, the melt growth rate is  $V_{\text{melt}} = B/\sqrt{t}$ , where  $B$  is an unknown constant to be obtained from fitting (related to mass balance). We numerically solve the diffusion–dissolution equation in semi-infinite space (eqs (A1) and (A2) in Appendix A) with concentration-dependent  $D$  given in eq 4. A nonlinear least-squares method based on Levenberg–Marquardt algorithm is used to minimize the sum of squares of errors along a concentration profile. The fitting yields values of  $D_{X=1}$  and  $\beta$  from which EBD of SiO<sub>2</sub> at various points including  $D_{\text{farfield}}$  and  $D_{\text{interface}}$  can be calculated. It turns out that the SiO<sub>2</sub> profile in each individual experiment does not necessarily constrain the parameters well due to correlated errors in  $\beta$  and  $D_{X=1}$ , though the profiles in basaltic melts provide better constraints owing to a larger compositional span. To improve the fitting, we fit all the concentration profiles (regardless of initial melt compositions and experimental durations) at the same experimental temperature (e.g., ~1300 °C) together to obtain a single  $\beta$  value while allowing  $D_{X=1}$  to vary from one experiment to another, because the slopes in the plot of  $\ln D$  versus either SiO<sub>2</sub> or  $X$  are almost the same at the same experimental temperature in rhyolite and basalt (Figure 6). The benefit of doing so is to best constrain the parameters of  $\beta$ . All fitting results are summarized in Table 4. SiO<sub>2</sub> diffusivities at intermediate melt compositions ( $X_{\text{Si+Al}} = 0.65, 0.7, 0.75, 0.8, 0.85, 0.9, \text{ and } 0.95$ ) are provided in Table 5. In addition, SiO<sub>2</sub> diffusivities along the melt composition profiles are provided in Supporting Information. A comparison

between the fit concentration profile and the measured data is illustrated in Figure 5 as the blue solid curve. Other fits are shown in Supporting Information. The  $D$  versus  $C$  relations obtained from the fits are shown in Figure 6 as lines (red dash lines for quartz dissolution in basalt, and blue dash lines for quartz dissolution in rhyolite), which are in good agreement with the numerical results using the Boltzmann analysis (Figure 6).

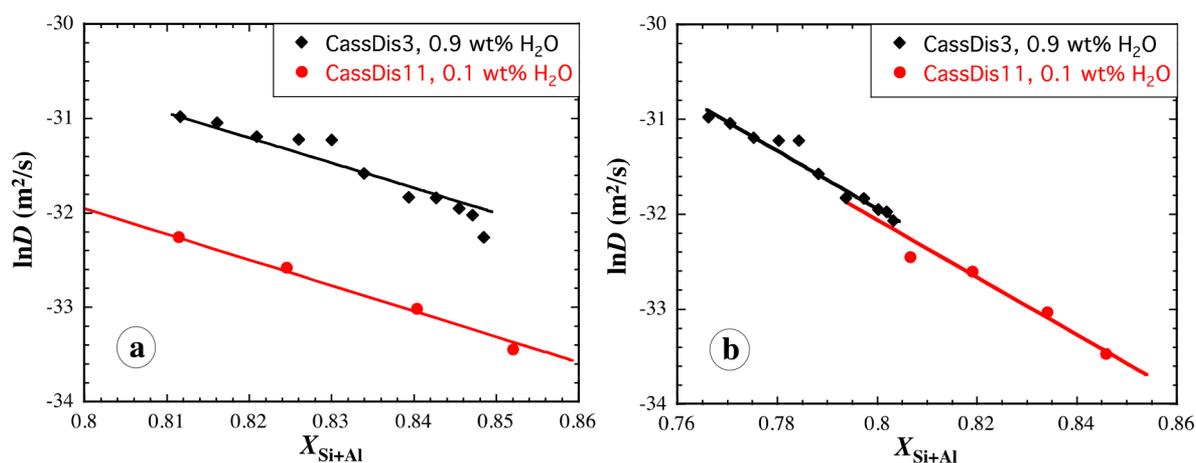
**4.1.3. Compositional and Temperature Dependence of SiO<sub>2</sub> Diffusivity in Silicate Melts.** In this study, the time-series experimental design and the wide range of experimental temperatures and melt composition enable us to assess the compositional dependence of  $D_{\text{SiO}_2}$  on both Si and Al, and to evaluate how it relates to the temperature. Fitting  $D_{\text{SiO}_2}$  values in Supporting Information Tables 1 and 2 (2046 points) as a function of  $X$  and  $T$  ( $X > 0.6$ ;  $T = 1300\text{--}1600$  °C) leads to

$$D_{\text{SiO}_2}^{\text{quartz dissolution}} = \exp\left(-14.168 + 2.758(1 - X) - \frac{35003 - 38829(1 - X)}{T}\right) \quad (5)$$

where  $X$  is the cation mole fraction of Si + Al and  $T$  is in K. The above equation reproduces our data (Supporting Information Tables S1 and S2) with  $1\sigma$  standard deviation of 0.32  $\ln D$  units and maximum deviation of 0.95  $\ln D$  units. Extrapolation for applications to 800 to 1200 °C will introduce additional errors. In calculating  $X$ , H in H<sub>2</sub>O (e.g., 0.32 wt % H<sub>2</sub>O in JDF) is included in the cations. Although eq 5 is applicable to rhyolitic to basaltic melts, calculation of  $X$  requires the full melt composition, which is sometimes unavailable or inconvenient. Hence, we also relate  $D_{\text{SiO}_2}$  for quartz dissolution in individual melts to SiO<sub>2</sub> wt % ( $C$ ) and  $T$  as follows



**Figure 6.**  $\text{SiO}_2$  diffusivity from Boltzmann analyses (points) and functional fitting results (dash line) as a function of  $\text{SiO}_2$  concentration (left column) and  $X$  (right column). The analyzed samples are quartz dissolution experiments in rhyolitic (Newberry) and basaltic (JDF) melts at 1300, 1400, 1500, and 1600 °C. One quartz dissolution experiment in andesitic melt at 1300 °C is from ref. 2.



**Figure 7.** Data points from Boltzmann analyses and functional fitting results (solid lines) of two cassiterite dissolution experiments in CIT and KS rhyolitic melts at 1000 °C. (a) Si and Al cation mole fractions are calculated based on the dry composition; that is, H<sub>2</sub>O was ignored when calculating X; (b) H<sub>2</sub>O was included by considering H as a cation when calculating X.

$$D_{\text{SiO}_2}^{\text{diss in rhyolite}} = \exp\left(-7.086 - 0.09469C - \frac{26109 + 28.77C}{T}\right) \quad (6a)$$

$$D_{\text{SiO}_2}^{\text{diss in basalt}} = \exp\left(-14.751 + 0.02051C - \frac{5052 + 280.5C}{T}\right) \quad (6b)$$

The maximum error in reproducing experimental data is 0.6 to 0.7 ln *D* units for both. That is, the equations for specific melt systems are more accurate than eq 5.

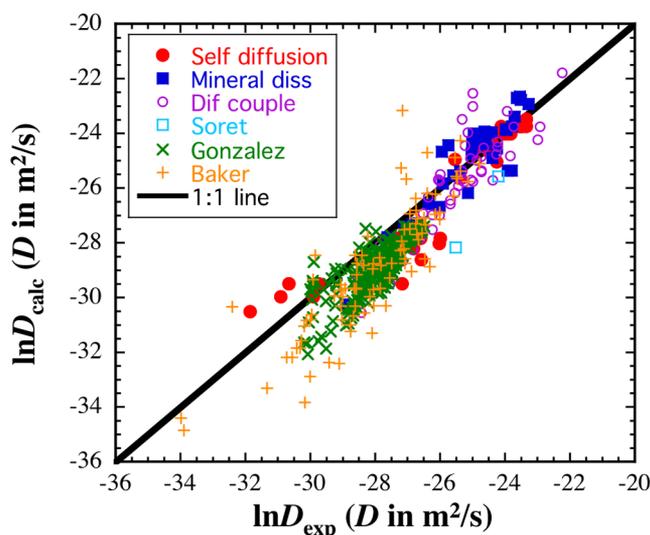
**4.1.4. Effect of H<sub>2</sub>O on SiO<sub>2</sub> Diffusivity.** Volatiles, especially H<sub>2</sub>O, have long been demonstrated to have significant and complex effects on diffusivities of other components.<sup>23–26,34,50,53</sup> The addition of H<sub>2</sub>O in silicate melts usually accelerates significantly the diffusion of other components. Such phenomena have often been attributed to the depolymerization of the melt structure by H<sub>2</sub>O. Quantitative attempts to model these effects have often treated H<sub>2</sub>O concentration as a separate parameter assuming various relations (such as linear, logarithmic, and square root) between *D* and H<sub>2</sub>O content.<sup>34</sup>

We used cassiterite dissolution experiments in various hydrous rhyolitic melts<sup>17</sup> to examine the effect of H<sub>2</sub>O on SiO<sub>2</sub> diffusivity. Two of the experiments (CassDis3 and CassDis11) were conducted at the same experimental temperature (1000 °C) in rhyolites with 0.9 wt % H<sub>2</sub>O and 0.1 wt % H<sub>2</sub>O. ln *D* versus *X* data are shown in Figure 7. When *X* is calculated based on dry glass composition, the ln *D* versus *X* trends of these two experiments clearly offset by about 1.4 ln *D* units (Figure 7a). However, when including H the same way as other cations (such as Fe, Ca, Na) in the calculation of *X*, the ln *D* versus *X* plots of the two experiments fall into the same trend within errors (Figure 7b). Hence, the effect of H<sub>2</sub>O on Si diffusivity appears to be the simple dilution of the mole fraction of network forming cations (Si and Al). The experiments at 5.9 wt % H<sub>2</sub>O would provide much better constraints but these experiments were at lower temperatures and only one experiment has barely resolvable SiO<sub>2</sub> concentration gradients, unable to provide strong constraints. Nonetheless, Zr diffusivity is also consistent with this observation to higher H<sub>2</sub>O contents.<sup>19</sup>

**4.1.5. Comparison with Literature Data.** Many authors investigated SiO<sub>2</sub> diffusion in silicate melts.<sup>1,21–26,31–34,50</sup> Here, we test our model (eq 5) using literature data. Note that our study reports effective binary diffusivities of SiO<sub>2</sub>.

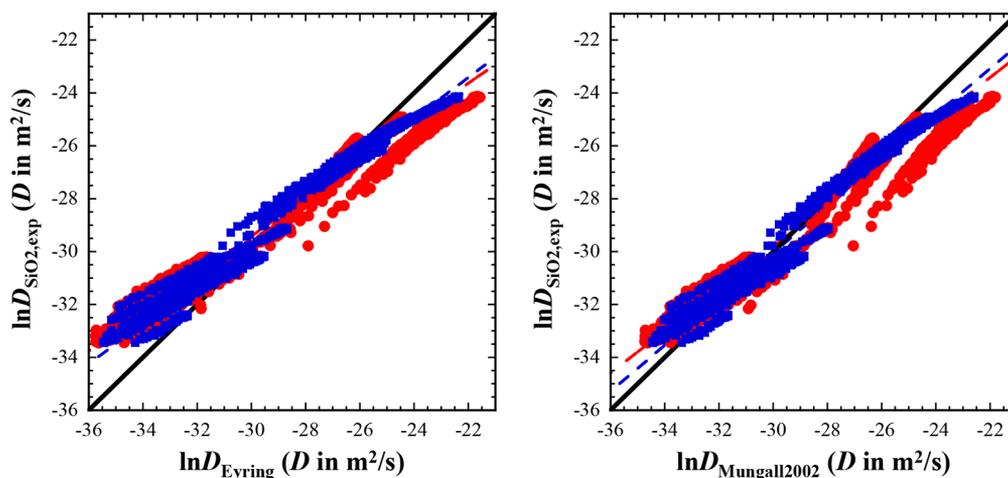
Because SiO<sub>2</sub> effective binary diffusivities can depend on concentration gradients even in the same bulk composition,<sup>20,45,46</sup> Equation 5 is best applied to SiO<sub>2</sub> diffusion during quartz dissolution. However, there are no literature data on SiO<sub>2</sub> diffusivities during quartz dissolution for direct comparison (e.g., ref 1 conducted quartz dissolution experiments but did not report SiO<sub>2</sub> diffusivities for these experiments; diffusion data during quartz dissolution in ref 2 are used in this study). Hence, we will use all Si diffusion data on silicate melts with composition close to natural melts for tests. Self- and chemical diffusion data for Si are reviewed in ref 34; data in the review are used but those on albite melt, diopside melt, CAS melt, and Na<sub>2</sub>Si<sub>4</sub>O<sub>9</sub> melt are excluded (the compositions of these melts are far from natural silicate melts). Data with 13 wt % H<sub>2</sub>O in ref 23 are excluded because the prediction is far off and because of uncertainty in the H<sub>2</sub>O concentration. The H<sub>2</sub>O content in the used data in Figure 8 ranges from 0 to 6.7 wt %. Additionally, recent SiO<sub>2</sub> effective binary diffusion data<sup>45,46,50</sup> are also included. Because eq 5 does not evaluate the pressure effect, data with *P* > 2 GPa are excluded. The comparison is shown in Figure 8. Equation 5 predicts Si self-diffusivities<sup>31,32,54</sup> to 0.11 ± 0.55 ln *D* units, the effective binary diffusivities during dissolution of olivine, diopside, rutile, spinel, and anorthite<sup>2,15,16,55,56</sup> to −0.02 ± 0.73 ln *D* units, the effective binary diffusion data of Gonzalez–Garcia et al.<sup>50</sup> from diffusion couple experiments to −0.97 ± 0.59 ln *D* units, the tracer and effective binary diffusivities of Baker and co-workers<sup>22,24–26,57,58</sup> to −0.90 ± 1.27 ln *D* units, and the effective binary diffusivities of other authors<sup>23,27,45,46,59,60</sup> to −0.34 ± 0.93 ln *D* units. Notably, Si effective binary diffusivities during mineral dissolution and self-diffusivities are well predicted. The largest discrepancy is for the diffusion data by Baker and co-workers. Some of the discrepancies may be attributed to (i) the effect of the halogens in their experiments, (ii) the use of one single parameter *X* (= Si + Al) to capture the compositional effect (including the effect of H<sub>2</sub>O) of complicated silicate melts, and (iii) the variable concentration gradients, leading to additional deviations.<sup>45,46</sup>

**4.1.6. Relation between SiO<sub>2</sub> Diffusivity and Melt Viscosity.** There is much expectation that SiO<sub>2</sub> diffusivity is related to melt viscosity. For example, SiO<sub>2</sub> tracer diffusivity might be the same as Eyring diffusivity.<sup>34,61</sup> Mungall<sup>62</sup>



**Figure 8.** Testing eq 5 using literature SiO<sub>2</sub> diffusion data. Data coverage: temperature between 900 and 1710 °C, pressure ≤ 2 GPa, H<sub>2</sub>O ≤ 6.7 wt %. Soret diffusion data are from ref 21. Gonzalez mean are from ref 50, whose data are plotted separately partially because the data set contained 242 points, more than all other data combined. Sources of other data can be found in the text.

proposed a relation between the tracer diffusivity of high-field strength elements and viscosity. Note that SiO<sub>2</sub> diffusivities obtained in this study are effective binary diffusivities, rather than tracer diffusivities. Figure 9 compares experimental SiO<sub>2</sub> EBD from this work to Eyring diffusivities and those by the model of Mungall.<sup>62</sup> In calculating the Eyring diffusivities, the viscosity models of refs 63 and 64 are used, and the jumping distance is assumed to be 0.28 nm. The choice of 0.28 nm is because Si tracer diffusivity in dry melts is often similar to oxygen tracer diffusivity<sup>34</sup> and 0.28 nm is the diameter of the O<sup>2-</sup> ion. The comparisons show that either the Eyring model or the Mungall model is not too far off with the largest difference no more than 1 order of magnitude. However, predicting SiO<sub>2</sub> diffusivity from viscosity is not as good as using eq 5.



**Figure 9.** Comparison of experimental SiO<sub>2</sub> EBD with Eyring diffusivity and with diffusivity calculated using the tracer diffusivity model for high-field strength elements by Mungall.<sup>62</sup> In calculating  $D$  values, the viscosity models of ref 63 (red circles) and ref 64 (blue squares) are used. The heavy solid line is 1:1 line.

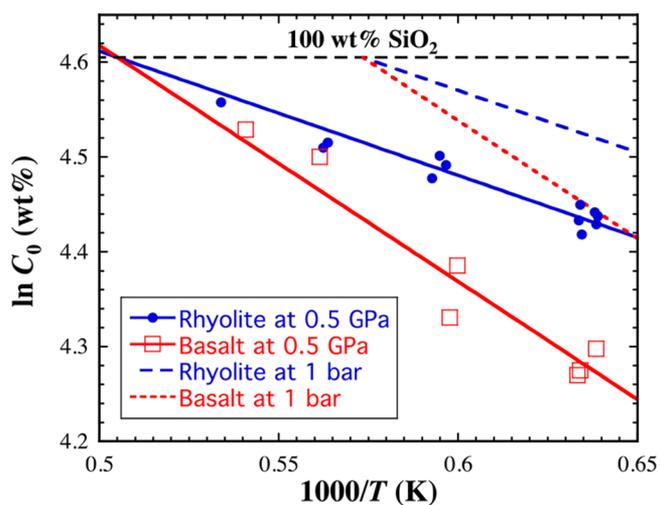
**4.2. SiO<sub>2</sub> Concentration at Quartz Saturation.** The interface SiO<sub>2</sub> concentration ( $C_0$ ) during quartz dissolution in every experiment is extracted from fitted profile as the extrapolated concentration at  $x = 0$  (Table 3). The interface SiO<sub>2</sub> concentrations at a given temperature and pressure are roughly independent of the experimental duration within experimental uncertainty (Figure 4a,b). That is, quartz dissolution is diffusion-controlled, and the interface melt composition can be approximately regarded as the equilibrium composition of the melt with quartz.

SiO<sub>2</sub> content plays the dominant role for quartz saturation, while other components play minor roles through their effect on the activity coefficient of SiO<sub>2</sub> in the melt. SiO<sub>2</sub> activity (equaling activity coefficient times concentration) in the melt equilibrated with quartz at the same temperature must be the same. However, SCQS (silica concentration at quartz saturation) is different between rhyolitic or basaltic melts at the same temperature and pressure (Table 3). For example, at 1300 °C and 0.5 GPa, SCQS is between 83.0 and 85.6 wt % (the range is regarded as experimental and extrapolation uncertainty) when the initial melt is rhyolite, but between 71.5 and 73.5 wt % if the initial melt is basalt. Note that the interface melts (saturation melts) in both cases are rhyolitic with the former containing significantly lower FeO, MgO, and CaO but higher K<sub>2</sub>O (Table 3). The higher SiO<sub>2</sub> concentration for quartz saturation in the former by  $(84.3/72.5 - 1) = 16\%$  than in the latter means that SiO<sub>2</sub> activity coefficient in the former interface melt (lower FeO, MgO, and CaO but significantly higher K<sub>2</sub>O) is smaller by 16%. The effect of other components on SiO<sub>2</sub> activity coefficient accounts for the difference in SCQS.

The rhyolite–MELTS software package<sup>12,13</sup> is used to calculate the liquidus based on the interface melt compositions we obtained. Note that these melts contain high SiO<sub>2</sub> concentrations up to 95 wt %, which are likely beyond the applicability range of the rhyolite–MELTS software. Given the interface melt composition, the rhyolite–MELTS software correctly predicts the liquidus phase to be quartz, but the calculated liquidus temperatures are higher than the experimental temperatures by 140 to 296 °C when the initial melt is rhyolitic, and by −92 to 142 °C when the initial melt

is basaltic. The large difference even when the interface  $\text{SiO}_2$  concentration is 71–85 wt % means that there is much room for improvement of the rhyolite–MELTS, and our data on SCQS at various temperatures (Table 3) can provide important constraints for improvement of thermodynamic models of silicate melts.<sup>9–13,65</sup>

Before such improvement is available, there is a need to develop a simple and practical way to approximate the quartz–melt equilibrium for the purpose of quantifying quartz dissolution and growth kinetics. We relate the  $\text{SiO}_2$  wt % concentration in the interface melt with the experimental temperature for different melt systems (Figure 10). Because



**Figure 10.**  $\text{SiO}_2$  concentration at  $\beta$ -quartz saturation (that is, interface  $\text{SiO}_2$  concentration, Table 3) versus  $1000/T$  with linear regression lines.  $\text{SiO}_2$  (100 wt %) is marked by the dashed horizontal line. Blue solid circles are for quartz dissolution in molten rhyolite at 0.5 GPa; red open squares are for quartz dissolution in molten basalt at 0.5 GPa. The blue long-dashed line is the estimated  $\ln C_0$  versus  $1000/T$  relation at 1 bar in rhyolite, and the red short-dashed line is that in basalt.

the interface  $\text{SiO}_2$  concentration ( $C_0$ ) is an apparent equilibrium constant, we express  $\ln C_0 = A - B/T$  where  $B \sim \Delta H/R$  with  $\Delta H$  being the standard enthalpy of the reaction  $\text{SiO}_2(\beta\text{-qz}) \rightleftharpoons \text{SiO}_2(\text{melt})$ , and  $R$  being the universal gas constant. Because we are using concentrations rather than chemical activities and because the activity coefficient depends on the concentrations of other components,  $A$  and  $B$  may vary from rhyolite to basalt. An important constraint is the melting temperature of  $\beta$ -quartz at 0.5 GPa (1707 °C, ref 39), meaning  $C_0 = 100\%$  at the temperature. Hence, we construct the  $\beta$ -quartz–melt equilibrium relation at 0.5 GPa for rhyolite and basalt respectively forcing each to go through  $C_0 = 100\%$  at 1980.15 K (solid blue and red lines in Figure 10)

$$\text{In rhyolite at 0.5 GPa: } \ln C_0 = 5.267 - 1310/T, \quad r^2 = 0.916, \quad (7a)$$

$$\text{In basalt at 0.5 GPa: } \ln C_0 = 5.862 - 2489/T, \quad r^2 = 0.933 \quad (7b)$$

where  $C_0$  is  $\text{SiO}_2$  concentration in wt % in the interface melt (that is, SCQS). When using the experimental temperature to predict the experimental interface  $\text{SiO}_2$  concentration, the largest error is 1.9 wt % using eq 7a (or 0.02 in  $\ln C_0$ ) when the initial melt is rhyolitic, and 3.4 wt % using eq 7b (or 0.04 in  $\ln C_0$ ) when the initial melt is basaltic. When using the

experimental interface  $\text{SiO}_2$  concentration to predict the experimental temperature, the largest error is 48 °C for both eq 7a when the initial melt is rhyolitic and eq 7b when the initial melt is basaltic. These prediction errors are significantly smaller than those from the calculation of the rhyolite–MELTS package (Gualda et al., 2012; Ghiorso et al., 2015).

Note that although the variation of diffusivity in silicate melts with pressure is often negligible for a pressure interval of 1 GPa (e.g., ref 15), the saturation concentration  $C_0$  depends on pressure more strongly than diffusivity.<sup>15</sup> For example, the metastable equilibrium temperature between  $\beta$ -quartz and pure  $\text{SiO}_2$  melt is 1707 °C at 0.5 GPa and 1471 °C at 1 bar (ref 39), varying by 236 °C in 0.5 GPa pressure interval. To estimate the  $\ln C_0$  versus  $1/T$  relation at 1 bar, we use the metastable equilibrium temperature between  $\beta$ -quartz and pure  $\text{SiO}_2$  melt at 1 bar, meaning that the line must pass through the point of (1000/1744.15,  $\ln 100$ ), and ignore the variation of  $B$  in eqs 7a and 7b in the pressure range of 1 bar to 0.5 GPa, leading to

$$\text{In rhyolite at 1 bar: } \ln C_0 \approx 5.356 - 1310/T \quad (7c)$$

$$\text{In basalt at 1 bar: } \ln C_0 \approx 6.032 - 2489/T \quad (7d)$$

These two rough equations are shown as dashed blue and red lines in Figure 10.

**4.3. Diffusive Mineral Dissolution Rates for Concentration-Dependent  $D$ .** For diffusive crystal dissolution when the diffusivity of the principal equilibrium-determining component is constant across the concentration profile, the melt growth distance can be expressed by the parabolic relation<sup>2</sup>

$$L_{\text{melt}} = 2\alpha\sqrt{Dt} \quad (8)$$

where  $\alpha$  is determined using eq 2 from the concentrations of the principal equilibrium-determining component in the interface melt ( $C_0$ ), far-field melt ( $C_\infty$ ), and the mineral ( $C_c$ ). In the case of quartz dissolution, the diffusivity of  $\text{SiO}_2$  (the equilibrium-determining component) is not constant. No theory is currently available to estimate diffusive crystal dissolution rate (or melt growth rate) when the diffusivity depends on concentration. For example, Yang et al.<sup>17</sup> showed that Sn diffusion during cassiterite dissolution is concentration dependent, but there were not enough data to develop a predictive theory for mineral dissolution rate or melt growth rate. This section uses an empirical approach to address how the melt growth distance or mineral dissolution distance can be estimated when  $D$  depends on concentration.

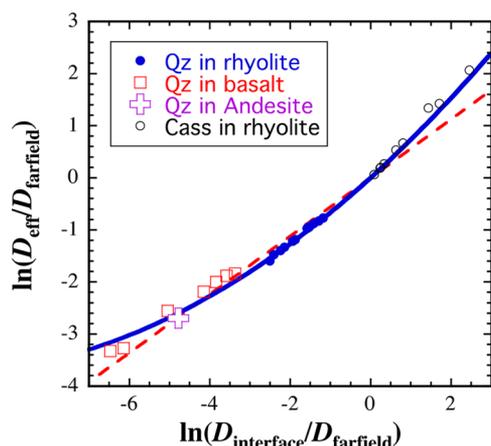
It has been shown using Boltzmann analysis that even when  $D$  depends on concentration, the concentration profile still propagates according to  $\sqrt{t}$ , and during mineral dissolution the melt growth distance is still proportional to  $\sqrt{t}$  (Appendix A). Experimental data are consistent with this expectation. For example, Figure 4c,d also shows that quartz dissolution distance is proportional to  $\sqrt{t}$ . In addition, even though  $\text{H}_2\text{O}$  diffusivity in silicate melts depends on its concentration,<sup>66,67</sup> dehydration mass loss when the far-field composition has not changed is also proportional to  $\sqrt{t}$  (refs 68 and 69). Hence, in modeling the melt growth distance during quartz dissolution in silicate melts, we adopt the formulation of eq 8 with  $\alpha$  solved from eq 2 but interpreting the constant  $D$  in the equation to be an effective diffusivity,  $D_{\text{eff}}$  leading to

$$L_{\text{melt}} = 2\alpha\sqrt{D_{\text{eff}}t} \quad (9)$$

Equation 9 leads to

$$D_{\text{eff}} = \frac{L_{\text{melt}}^2}{4\alpha^2 t} \quad (10)$$

$D_{\text{eff}}$  values so calculated are listed in Table 4. Because  $D_{\text{eff}}$  is an effective diffusivity across the entire concentration profile, one might expect that it may be expressed as a combination of the diffusivities at the interface melt and the far-field melt. In Figure 11,  $\ln(D_{\text{eff}}/D_{\text{farfield}})$  versus  $\ln(D_{\text{interface}}/D_{\text{farfield}})$  are



**Figure 11.** Relating effective diffusivity ( $D_{\text{eff}}$ ) for mineral dissolution to diffusivity at the far-field and interface melts. Quartz dissolution data in molten rhyolite and basalt are from this work. Quartz dissolution experiment in molten andesite is from ref 2. Cassiterite dissolution experiments in molten rhyolite are from ref 17. The red dashed line is a linear fit passing through the origin (eq 11); and the blue solid curve is a second order polynomial fit passing through the origin (eq 12).

plotted, including  $\text{SiO}_2$  diffusivity data in Table 4,  $\text{SiO}_2$  diffusivity obtained from exp #234 of ref 2 using concentration-dependent  $D_{\text{SiO}_2}$  (eq 4), and Sn diffusivity data during cassiterite dissolution in rhyolitic melt.<sup>17</sup> For the case of cassiterite dissolution, Sn rather than  $\text{SiO}_2$  diffusion data are used because Sn is the principal equilibrium-determining component for cassiterite saturation.

The data in Figure 11 show some curvature. For simplicity, a first order linear fit with an intercept of zero (red dashed line) is carried out to obtain  $k = 0.560 \pm 0.015$  with  $r^2 = 0.9708$

$$D_{\text{eff}} \approx (D_{\text{interface}})^{0.56} (D_{\text{farfield}})^{0.44} \quad (11)$$

To account for the curvature, a curve fit (blue solid curve in Figure 11) is obtained by forcing the intercept to be zero

$$\ln \frac{D_{\text{eff}}}{D_{\text{farfield}}} = \left( 0.6996 + 0.0327 \cdot \ln \frac{D_{\text{interface}}}{D_{\text{farfield}}} \right) \ln \frac{D_{\text{interface}}}{D_{\text{farfield}}} \quad (12)$$

with  $r^2 = 0.9927$ . In both fits, the intercept is forced to be zero because  $\ln(D_{\text{interface}}/D_{\text{farfield}}) = 0$  means constant  $D$ , indicating that  $\ln(D_{\text{eff}}/D_{\text{farfield}})$  must be zero. Work is currently in progress through systematic numerical simulations to derive a general relation for predicting diffusive mineral dissolution or growth rate when the diffusivity is concentration dependent.<sup>70</sup>

Once  $D_{\text{eff}}$  is predicted, the melt growth distance  $L$  during quartz and cassiterite dissolution may be calculated using eq 9, with  $\alpha$  from eq 2 and  $t$  being the experimental duration. When  $\alpha$  is positive, the melt grows (the crystal dissolves). When  $\alpha$  is negative, the melt is consumed, and the crystal grows. Worked numerical examples are given in the supplementary files to help readers to use the method.

**4.4. Convective Dissolution Rates.** For modeling convective dissolution or growth rates of a rising bubble or sinking/rising crystal when diffusivity varies significantly, as a first order approximation, we suggest using similar approaches as in refs 71–73 but replacing the constant  $D$  by  $D_{\text{eff}}$  and the boundary layer thickness by e-folding thickness  $\delta_{\text{eff}}$  (the distance in the melt from the mineral–melt interface at which the concentration equals  $C_0/e + C_\infty(1 - 1/e)$  where  $e = 2.71828\dots$ ). The use of e-folding distance avoids the need to find the very steep slope of the concentration profile at the mineral–melt interface. Hence, the general equation for predicting convective melt growth rate<sup>2</sup> becomes

$$V = \frac{D_{\text{eff}} (C_0 - C_\infty)}{\delta_{\text{eff}} (C_c - C_0)} \quad (13)$$

Watson<sup>1</sup> investigated dynamic quartz dissolution in a basaltic melt (containing 49.5 wt %  $\text{SiO}_2$ ) at 1300 and 1400 °C and 1 bar by rotating a cage with a quartz crystal in a bucket of melt at three rotations per minute. He measured oxide concentration profiles and found extremely steep concentration slope near the interface. He estimated the boundary layer thickness to be 75  $\mu\text{m}$ . He also estimated quartz dissolution rate of  $1.5 \times 10^{-6} \text{ g/cm}^2/\text{s} = 0.0059 \mu\text{m/s}$  at 1300 °C and  $3.3 \times 10^{-6} \text{ g/cm}^2/\text{s} = 0.013 \mu\text{m/s}$  at 1400 °C. This is a case of quartz dissolution with forced convection. The  $\text{SiO}_2$  concentration in the interface melt at 1300 °C and 1 bar is very roughly estimated from eq 7d to be 85.61 wt %, which is between the extrapolated values of the three experiments at 1300 °C (ref 1). Hence,  $(C_0 - C_\infty)/(C_c - C_0) = (85.61 - 49.5)/(100 - 85.61) \approx 2.51$ . Because diffusivities are not expected to change much in a 0.5 GPa interval,<sup>15,16</sup> we use eq 6b and obtain  $D_{\text{farfield}} = 6.41 \mu\text{m}^2/\text{s}$ , and  $D_{\text{interface}} = 0.0215 \mu\text{m}^2/\text{s}$ . Note the huge difference in  $D_{\text{farfield}}$  and  $D_{\text{interface}}$  by a factor of 300, consistent with the extremely steep slope near the interface in ref 1. Estimated  $D_{\text{eff}} = 0.344 \mu\text{m}^2/\text{s}$  using eq 12. The predicted melt growth rate using eq 13 is  $0.0115 \mu\text{m/s}$ , and quartz dissolution rate is  $0.0126 \mu\text{m/s}$ . Experimental quartz dissolution rate at 1300 °C is  $0.0059 \mu\text{m/s}$  (ref 1). The predicted value is about 2 times the observed value. Considering the roughness in estimating  $C_0$  and  $\delta$ , this level of agreement is deemed acceptable.

## 5. CONCLUSIONS

1. Quartz dissolution kinetics in rhyolitic and basaltic melts is complicated because the  $\text{SiO}_2$  concentration profile cannot be described by a constant diffusivity. We assessed the dependence of  $\text{SiO}_2$  diffusivity as a function of melt composition (including  $\text{H}_2\text{O}$  concentration) and temperature and developed a method to predict mineral dissolution rate for the case of concentration-dependent diffusivity.

2. The compositional dependence of  $D_{\text{SiO}_2}$ , including the effect of  $\text{H}_2\text{O}$ , can be roughly captured by an exponential dependence of  $D_{\text{SiO}_2}$  on the cation mole fraction of Si + Al. The dependence of  $D_{\text{SiO}_2}$  on Si + Al becomes weaker as

temperature increases.  $D_{\text{SiO}_2}$  during quartz dissolution in molten rhyolite, andesite and basalt can be roughly accounted for using a simple model (eq 5) with maximum error of 0.95 ln  $D$  units. The effects of other components are small and cannot be resolved yet.  $D_{\text{SiO}_2}$  is related to viscosity but using viscosity to predict  $D_{\text{SiO}_2}$  has larger errors than using eq 5.

3. We developed a method to predict diffusive mineral dissolution distance in melts when the diffusivity is concentration-dependent. The mineral dissolution distance can be estimated using  $L = 2\alpha\sqrt{D_{\text{eff}}t}(\rho_{\text{melt}}/\rho_{\text{mineral}})$  where  $D_{\text{eff}}$  is the effective diffusivity of the principal equilibrium-determining component across the concentration profile and can be estimated from  $D$  in the far-field and interface melts. The approach works well for both quartz dissolution and cassiterite dissolution.

4. For predicting convective mineral dissolution rate, we propose to use  $D_{\text{eff}}$  to replace the constant  $D$  in existing literature treatments.

5. We derived the equation for using Boltzmann analysis to treat concentration-dependent diffusivity from diffusion profiles generated by diffusive crystal dissolution.

## APPENDIX A

### Boltzmann Analysis for Diffusion during Crystal Dissolution

Diffusion equation of mineral dissolution in the melt in the interface-fixed reference frame can be written as

$$\frac{\partial C}{\partial t} = \frac{\partial}{\partial x} \left( D \frac{\partial C}{\partial x} \right) - V \frac{\partial C}{\partial x} \quad (\text{A1})$$

with initial and boundary conditions

$$C|_{t=0} = C_{\infty}$$

$$\left( D \frac{\partial C}{\partial x} \right) \Big|_{x=0} + V(C_c - C_{x=0}) = 0 \quad (\text{A2})$$

Using Boltzmann transformation

$$\eta = \frac{x}{\sqrt{t}}; \quad \frac{\partial \eta}{\partial x} = \frac{1}{\sqrt{t}}; \quad \frac{\partial \eta}{\partial t} = -\frac{\eta}{2t} \quad (\text{A3})$$

Transform the diffusion equation using  $\eta$

$$-\frac{\eta}{2t} \frac{\partial C}{\partial \eta} = \frac{1}{t} \left[ \frac{\partial}{\partial \eta} \left( D \frac{\partial C}{\partial \eta} \right) \right] - \frac{V}{\sqrt{t}} \frac{\partial C}{\partial \eta} \quad (\text{A4})$$

$$V(C_c - C_{x=0}) + \frac{1}{\sqrt{t}} \left( D \frac{\partial C}{\partial \eta} \right) \Big|_{\eta=0} = 0 \quad (\text{A5})$$

When diffusion control is reached,  $C|_{x=0}$  is a constant, and  $D\partial C/\partial \eta$  at  $\eta = 0$  is a constant. Hence,  $V$  is inversely proportional to  $\sqrt{t}$  and may be written as  $B/\sqrt{t}$ .

Substitute  $V$  from eq A5 into the diffusion eq A4

$$-\frac{\eta}{2} \frac{\partial C}{\partial \eta} = \left( \frac{\partial}{\partial \eta} \left( D \frac{\partial C}{\partial \eta} \right) \right) + \frac{\left( D \frac{\partial C}{\partial \eta} \right) \Big|_{\eta=0}}{C_c - C_{\eta=0}} \frac{\partial C}{\partial \eta} \quad (\text{A6})$$

Rearrange the above equation (note that  $D|_{\eta=0}$  is a constant)

$$\left[ \frac{\left( D \frac{\partial C}{\partial \eta} \right) \Big|_{\eta=0}}{C_c - C_{\eta=0}} + \frac{\eta}{2} \right] \frac{\partial C}{\partial \eta} + \left[ \frac{\partial}{\partial \eta} \left( D \frac{\partial C}{\partial \eta} \right) \right] = 0 \quad (\text{A7})$$

Integrate the above differential equation along  $\eta$  from  $\eta_0$  to  $\infty$  and recognizing  $(\partial C/\partial \eta)|_{\eta=\infty} = 0$

$$\frac{\left( D \frac{\partial C}{\partial \eta} \right) \Big|_{\eta=0}}{C_c - C_{\eta=0}} (C_{\infty} - C_{\eta_x}) + \frac{1}{2} \int_{C_{\eta_x}}^{C_{\infty}} \eta \, dC - D \frac{\partial C}{\partial \eta} \Big|_{\eta_x} = 0 \quad (\text{A8})$$

Solving  $D$  from the above equation leads to

$$D = \frac{\frac{\left( D \frac{\partial C}{\partial \eta} \right) \Big|_{\eta=0}}{C_c - C_{\eta=0}} (C_{\infty} - C_{\eta_x}) + \frac{1}{2} \int_{C_{\eta_x}}^{C_{\infty}} \eta \, dC}{\frac{\partial C}{\partial \eta} \Big|_{\eta_x}} = \frac{\left( D \frac{\partial C}{\partial \eta} \right) \Big|_{x=0} \frac{(C_{\infty} - C_x)}{C_c - C_{x=0}} + \frac{1}{2t} \int_{C_x}^{C_{\infty}} x \, dC}{\frac{\partial C}{\partial x} \Big|_x} \quad (\text{A9})$$

When  $x = 0$ ,  $D = D|_{x=0}$ , meaning

$$D_{x=0} = \frac{C_c - C_0}{C_c - C_{\infty}} \frac{\int_{C_0}^{C_{\infty}} x \, dC}{2t \frac{\partial C}{\partial x} \Big|_{x=0}} \quad (\text{A10})$$

Substitute  $D|_{x=0}$  into eq A9

$$D = \frac{\frac{C_{\infty} - C_x}{C_c - C_{\infty}} \int_{C_0}^{C_{\infty}} x \, dC + \int_{C_x}^{C_{\infty}} x \, dC}{2t \frac{\partial C}{\partial x} \Big|_x}, \quad t > 0 \quad (\text{A11})$$

Hence, to calculate  $D$  at a given point  $x$ , we need  $C_x$ ,  $(\partial C/\partial x)_x$ ,  $\int x \, dC$  from  $C_x$  to  $C_{\infty}$ . Other parameters in the above equation are constants for a given experiment and a given profile.

## ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsearthspacechem.8b00193.

Appendix B including worked numerical examples for calculating diffusive dissolution distance; MatLab code for carrying out Boltzmann analysis; SiO<sub>2</sub> concentration profile with fits in every experiment (PDF)

Electron microprobe data for quartz dissolution experiments in rhyolite (XLSX)

Electron microprobe data for quartz dissolution experiments in basalt (XLSX)

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### Notes

The authors declare no competing financial interest.

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# Kinetics of Quartz Dissolution in Natural Silicate Melts, and Dependence of SiO<sub>2</sub> Diffusivity on Melt Composition

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## Appendix B. Supporting information

1. Worked numerical examples for calculating diffusive dissolution distance when  $D$  depends on concentration
2. MatLab code for carrying out Boltzmann analysis to diffusion profiles during mineral dissolution
3. SiO<sub>2</sub> concentration profile with fits in every experiment
4. Supplementary Table 1:  
Electron microprobe data for quartz dissolution experiments in rhyolite
5. Supplementary Table 2:  
Electron microprobe data for quartz dissolution experiments in basalt

### 1. Worked numerical examples for calculating diffusive dissolution distance when $D$ depends on concentration

*Example 1.* Consider quartz dissolution experiment QzDisBa101 (this study) at 1293°C for 3667 s in a basaltic melt initially containing 49.9 wt% SiO<sub>2</sub>. Calculate the melt growth distance and compare with experimental melt growth distance  $L_{\text{melt}} = 29.6 \mu\text{m}$  (Table 2).

[*Solution*] First we estimate  $C_0$  using Eq. (7b) to be 71.72 wt%. Note that  $C_\infty = 49.9$  wt% (SiO<sub>2</sub> concentration in the initial melt), and  $C_c \approx 100$  wt% (SiO<sub>2</sub> concentration in quartz). Hence,  $(C_0 - C_\infty)/(C_c - C_0) = (71.5 - 49.9)/(100 - 71.5) = 0.7715$ . Solving Eq. (2) leads to  $\alpha = 0.2996$  (equation # means equation in the main text of this work unless otherwise indicated). Using Eq. (6b),  $D_{\text{farfield}} = 5.70 \mu\text{m}^2/\text{s}$ , and  $D_{\text{interface}} = 0.179 \mu\text{m}^2/\text{s}$ . The variation of  $D_{\text{SiO}_2}$  along the concentration profile is a factor of 32. Using Eq. (11), we predict  $D_{\text{eff}} \approx 0.821 \mu\text{m}^2/\text{s}$ , and  $L_{\text{melt}} = 33 \mu\text{m}$ . Using Eq. (12), we predict  $D_{\text{eff}} \approx 0.749 \mu\text{m}^2/\text{s}$ , and  $L_{\text{melt}} = 31 \mu\text{m}$ . Both of these predictions are in good agreement with the observed value of 29.6  $\mu\text{m}$ .

*Example 2.* Consider cassiterite dissolution experiment CassDis1 (Yang et al., 2016) at 1100°C for 1856 s in a rhyolitic melt with negligible initial Sn concentration. The interface SnO<sub>t</sub> concentration is 12.831 wt%. Calculate the melt growth distance and compare with experimental melt growth distance  $L_{\text{melt}} = 2.6 \mu\text{m}$  (Table 2).

[*Solution*] First we calculate  $(C_0 - C_\infty)/(C_c - C_0) = 0.1676$  where  $C$  is concentration of SnO<sub>t</sub>, with  $C_\infty = 0$ ,  $C_0 = 12.831$  wt%, and  $C_c = 100$  wt% SnO<sub>2</sub> = 89.382 wt% SnO<sub>t</sub>. Solving Eq. (2) for

$\alpha$  leads to:  $\alpha = 0.0856$ . From Yang et al. (2016),  $D_{\text{farfield}} = 0.0225 \mu\text{m}^2/\text{s}$  (Eq. 7 in Yang et al., 2016), and  $D_{\text{interface}} = 0.158 \mu\text{m}^2/\text{s}$  using  $D_{\text{Sn}} = D_{\text{farfield}}\exp(0.1561C)$  in Yang et al. (2016). Then, using Eq. (11), we predict  $D_{\text{eff}} \approx 0.067 \mu\text{m}^2/\text{s}$ , and  $L_{\text{melt}} = 1.9 \mu\text{m}$ . Using Eq. (12), we predict  $D_{\text{eff}} \approx 0.100 \mu\text{m}^2/\text{s}$ , and  $L_{\text{melt}} = 2.3 \mu\text{m}$ , fairly close to the observed value.

Note that in the above two examples, at least three uncertainties, that in predicting the interface melt composition, that in the diffusivity expression (such as Eq. (6)), and that in Eq. (12), all contribute to the error in the calculated melt growth distance. Hence, the level of agreement (within about 20% relative) in the above examples is deemed very good.

## 2. MatLab code for carrying out Boltzmann analysis

```
%% Main program; Boltzmann analysis of Qz dissolution profile
% Yu et al., 2019, ACS Earth & Space Chemistry
% Description; Main
% Diffusivity of SiO2 depends on concentration, varying across the profile
clear;
figure_clean();
%% Sample Number & CompNum
% Set ExpNum as 1+real ExpNum, e.g. if ExpNum is 110; then the value is set as 1110.
ExpNum = 1310;
% For quartz dissolution set CompNum = [1, 2]; for cassiterite set CompNum = [1, 3]
CompNum = [1, 2];
dx = 5; %μm
%smoothing segments
smooth_seg = [50,100];
%number of averaged points [10,25,25]
smooth_factor = [10, 25, 25];
%% load data
[Profile, ExpCond, Err] = load_data(ExpNum, CompNum);
% check loading error
if cell2mat(Err(1))
    err_info(Err);
    return;
end
%experimental conditions
[Duration, Cinf, Cc, T] = exp_condition(ExpCond );
%fit the SiO2 and X(Si+Al) relation: 2nd order polynomial
[cxfcn, gof, op] = fit(Profile(:,2), Profile(:,3), 'poly2');
Coefs = coeffvalues(cxfcn);
fc = @(x) Coefs(1)*x.^2 + Coefs(2)*x + Coefs(3);
end
%% Smooth the data & Display the smoothed profile
ProfileSmooth = profile_smoother(Profile(:,1:2), smooth_factor,smooth_seg, dx, Cinf,
ExpNum);
%% Boltzmann Analysis & Display the relation between D and C
[D_C, D_C75, V, A] = boltzmann(ProfileSmooth, Duration, Cc', Cinf', dx, ExpNum, fc);
%% export the result into files
OpFileName = strcat('./output/Boltzmann-SiO2-', num2str(ExpNum), '.csv');
OpFile = fopen(OpFileName, 'w');
% OpContent =
[xdata(NumPoints_col(i)+1:NumPoints_col(i+1)),xdata(NumPoints_col(i)+1:NumPoints_col(i
+1))*sqrt(Duration(i)), Profile(NumPoints_col(i)+1:NumPoints_col(i+1), 1),
Profile_All(NumPoints_col(i)+1:NumPoints_col(i+1),:)]';
fprintf(OpFile, '%10s,%8s,%8s\n', 'SiO2 (wt%)', 'X(Si+Al)', 'lnD');
fprintf(OpFile, '%10.5f,%8.5f,%8.4f\n', D_C75');
fclose(OpFile);

function [] = figure_clean(n)
% Yu et al., 2019, ACS Earth & Space Chemistry
%figure_clean()
%Description
%When opened figures exceed the number of n (default, 3), close all figures
h = findobj('type', 'figure');
if nargin < 1
    n = 3;
end
if n == 0
    return;
else
    length(h) >= n;
```

```

    close all;
end
end

```

```

function [Profile, ExpCond, Err] = load_data(ExpNum, CompNum)
% Yu et al., 2019, ACS Earth & Space Chemistry
%[Profile, ExpCond, Err] = load_data(ExpNum, CompNum)
%Description
%load the basic data of each experiment from csv file
%load experiment conditions: T, P, duration, farfield melt comp
ExpCond = importdata('./data/Data_ExpCond.csv');
ExpCond = ExpCond.data;
ExpIdx = find(ismember(ExpCond(1,:), ExpNum));
%No sample found, return error 1
if isempty(ExpIdx)
    Err = {1, 'No such sample found'};
    Profile=[];
    ExpCond=[];
    return;
end
ExpCond = ExpCond(:,ExpIdx);
%load the concentration profiles
ProfileName = strcat('./data/Data-Profile-',num2str(ExpNum),'.csv');
Profile = importdata(ProfileName);
Profile=Profile.data;
Profile = Profile(:,[2,2+CompNum]);
Err = {0,'data loaded successfully'};
end

```

```

function [] = err_info(Err)
% Yu et al., 2019, ACS Earth & Space Chemistry
%err_info(Err)
% report the error information regarding the different situations
fprintf(char(Err(2)));
fprintf('\n');
end

```

```

function [Duration, Cinf, Cc, T] = exp_condition(ExpCond, CompNum)
% Yu et al., 2019, ACS Earth & Space Chemistry
%[Duration, Cinf, Cc,T] = exp_condition(ExpCond)
%Description
%load the experimental duration, Qz SiO2 comp, and farfield comp
Duration = ExpCond(4);
Cinf = ExpCond(5);
Cc = ExpCond(7);
T = ExpCond(2)
end

```

```

function [ProfileSmooth] = profile_smoother(Profile, smooth_factor, smooth_seg, dx,
Cinf, ExpNum, CompNum)
% Yu et al., 2019, ACS Earth & Space Chemistry
%[Profile_s] = profile_smoother(Profile, smooth_factor, ExpNum)
%Description; Smooth the diffusion profiles without interpolation

```

```

x = Profile(:,1);
[smooth_x, ind] = unique(x);
NumPoints = ceil(max(x)/dx);
ProfileSmooth = zeros(NumPoints+2, size(Profile,2));
ProfileLen = NumPoints*dx;
% for i = 1:length(CompNum)
    y1 = smooth(x, Profile(:, 2), smooth_factor(1));
    y2 = smooth(x, Profile(:, 2), smooth_factor(2));
    y3 = smooth(x, Profile(:, 2), smooth_factor(3));
    smooth_y1 = y1(ind);
    smooth_y2 = y2(ind);
    smooth_y3 = y3(ind);
smooth_y = [smooth_y1(smooth_x<=smooth_seg(1));
smooth_y2((smooth_x>smooth_seg(1)&(smooth_x<=smooth_seg(2)));smooth_y3(smooth_x>smooth_seg(2))];
ProfileSmooth(:,2) = interp1(smooth_x, smooth_y, (-dx:dx:ProfileLen)',
'linear','extrap');
ProfileSmooth(:,1) = (-dx:dx:ProfileLen)';
%% visualize the smoothed profile
figure('name', strcat('Smooth Profile: ', num2str(ExpNum)));
plot(ProfileSmooth(:,1), ProfileSmooth(:,2));
hold on;
scatter(Profile(:,1), Profile(:,2));
plot([0, ProfileLen], [Cinf(1), Cinf(1)], '--r');
hold off;
xlim([0, max(x)]);
title('SiO_{2}');
% end

```

```

function [D_C,D_C75, V, A] = boltzmann(Profile, Duration, Cc, Cinf, dx, ExpNum, fc)
% Yu et al., 2019, ACS Earth & Space Chemistry
%[D, V, A] = boltzmann(Profile, Duration, Cc, ExpNum, CompNum)
%Boltzmann analysis of diffusion profile, instantaneous dissolution
%rate is proportional to inverse of square root of time
%find the slope at all the points
dC_dx = diff_central(Profile(:,2), dx); %using smoothed profile
ProfileLen = length(dC_dx);
D = zeros(ProfileLen, 1);
% calculate D0
% Area_D0 = -(trapz(Profile(2:end-1,1), Profile(2:end-1,2:end),1)-Profile(end-1,2:end))*(ProfileLen-1)*dx);
Area_D0 = -(trapz(Profile(2:end-1,1), Profile(2:end-1,2:end),1)-Cinf*(ProfileLen-1)*dx);
% D0 = Area_D0.*((Cc-Profile(2,2:end))./(Cc-Profile(end-1,2:end)))./(2*Duration*dC_dx(1,:));
D0 = Area_D0.*((Cc-Profile(2,2))./(Cc-Cinf))./(2*Duration*dC_dx(1,1));
% calculate D at different C
for i = 1:ProfileLen
    %rectangle + erf curve - rectangle
    Area_i = -(Profile(i+1,2)*(i-1)*dx + trapz(Profile(1+i:end-1,1), Profile(1+i:end-1,2),1) - (ProfileLen-1)*dx.*Cinf);
    D(i,1) = (Area_i + Area_D0.*(Cinf - Profile(1+i,2))./(Cc - Cinf))./(2*Duration*dC_dx(i,1));
end
D = D*(1e-12);
%instantaneous interface reaction rate
V = -D0.*dC_dx(1,1)./(Cc-Profile(2,2));
A = V*sqrt(Duration);
D_C = [Profile(2:end-1,2), D];
%% visualize the result
D_logic = (D(:,1) > 0); %% (D(:,2) > 0) & (D(:,3) > 0) & (D(:,4) > 0);
% C75_logic = Profile(2:end-1, 2) > 51;

```

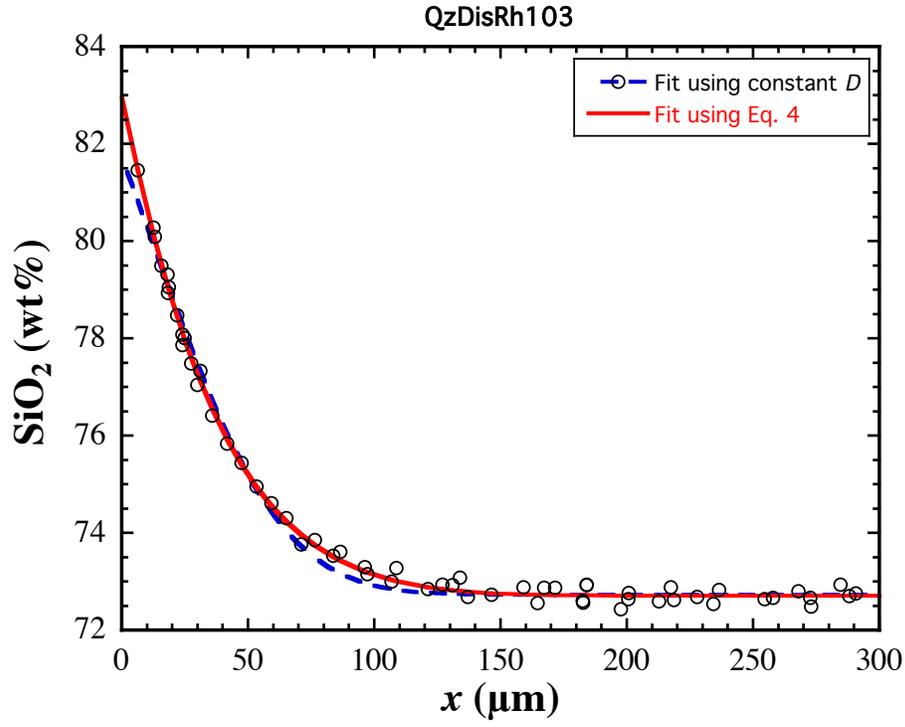
```

% D_logic = (prod(D_logic*1, 2)>0)&(C75_logic);
% D_temp = D.*D_logic;
D_keep = log(D(D_logic,1));
Profile_keep = Profile(2:end-1,2);
Profile_keep = Profile_keep(D_logic,1);
Profile_keep(:,2) = fc(Profile_keep(:,1));
D_C75 = [Profile_keep, D_keep];
% D_C75 = reshape(D_C75, size(Profile_keep,1), size(Profile_keep,2)*2);
figure('name', strcat('Boltzmann Analysis: ', num2str(ExpNum)));
for j = 1:2
    subplot(2,1,j);
    scatter(Profile_keep(:,j), D_keep(:,1),'xb');
    x_limit = [min(Profile_keep(:,j)), max(Profile_keep(:,j))];
%     xlim(x_limit);
    switch j
        case 1
            title('SiO_{2}');
        case 2
            title('Si+Al mol');
    end
    %for SiO2 reverse the x axis
    set(gca, 'XDir', 'reverse');
end
end

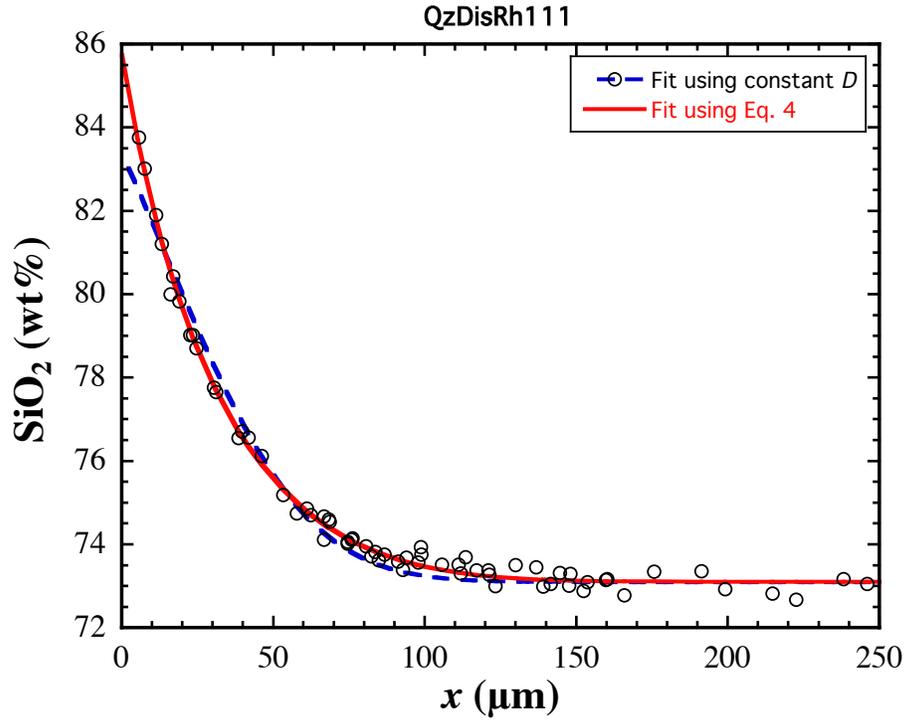
```

Supplementary materials – fit results of SiO<sub>2</sub> concentration profiles

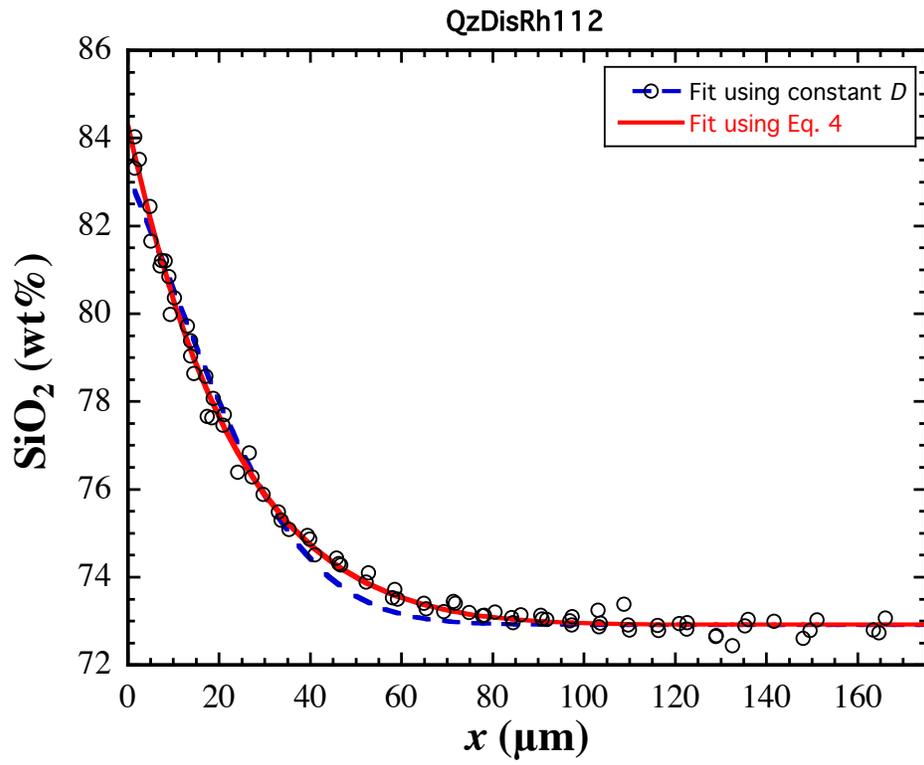
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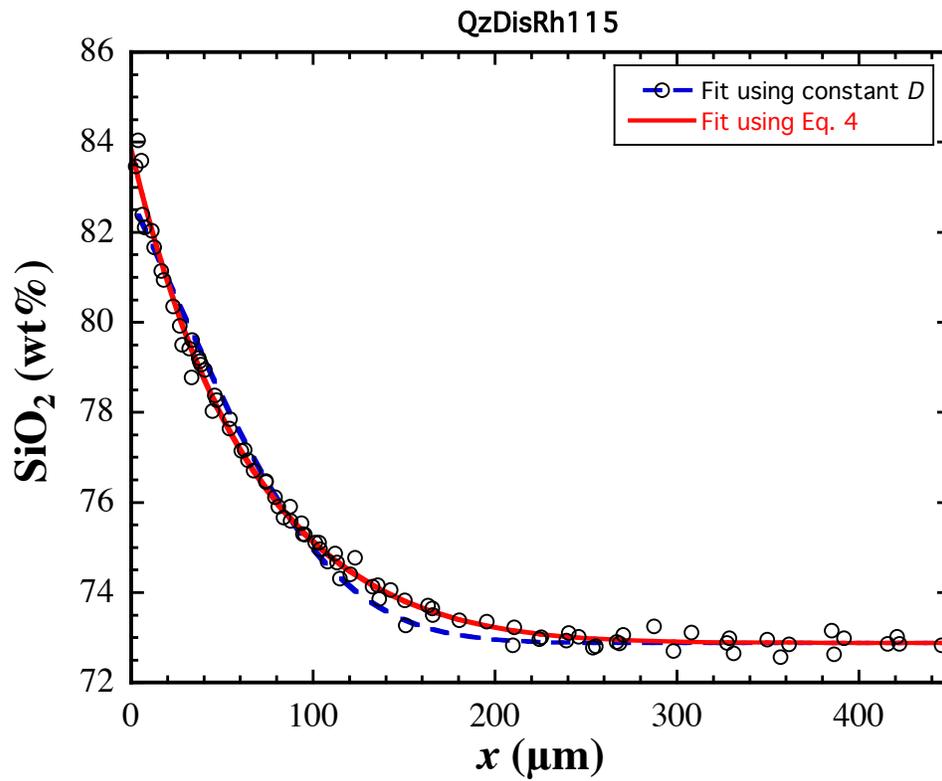
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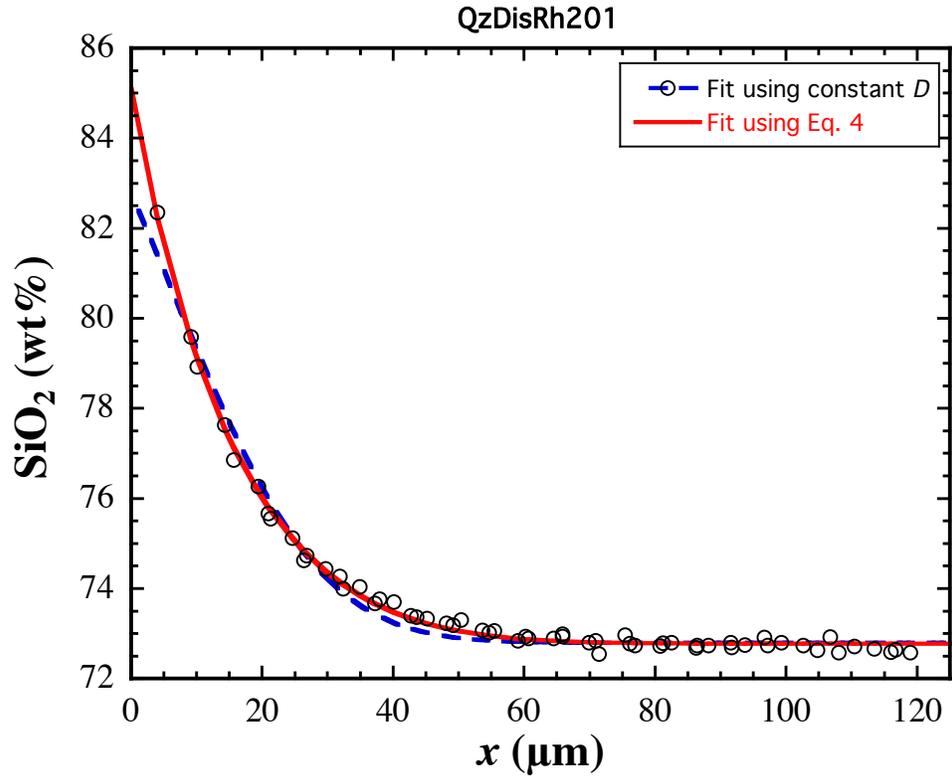
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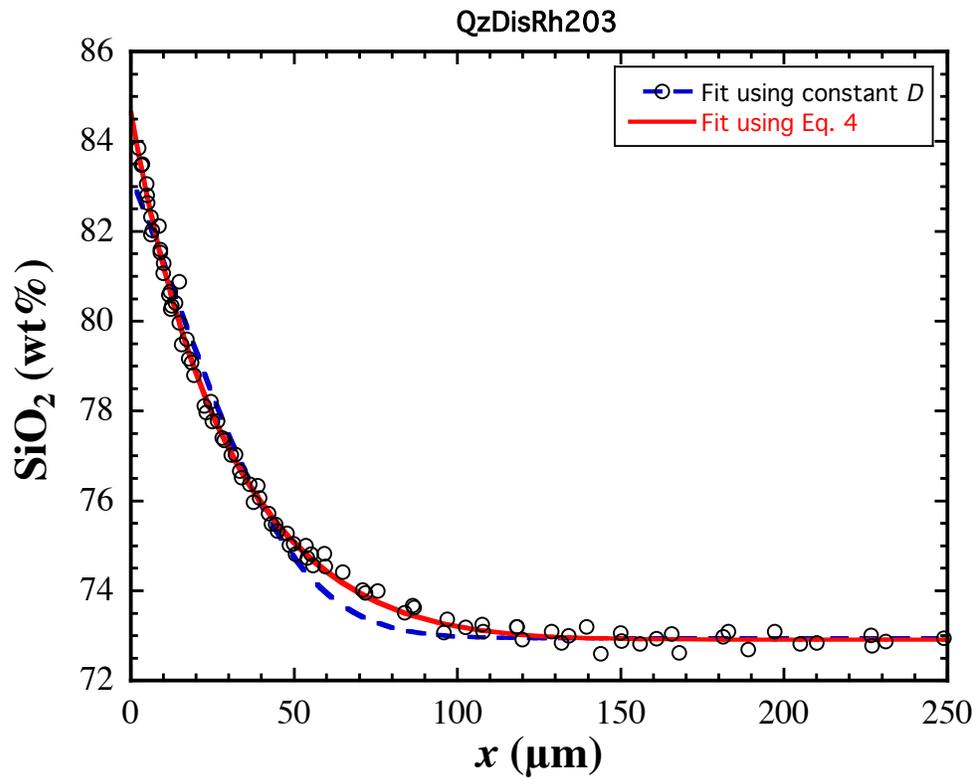
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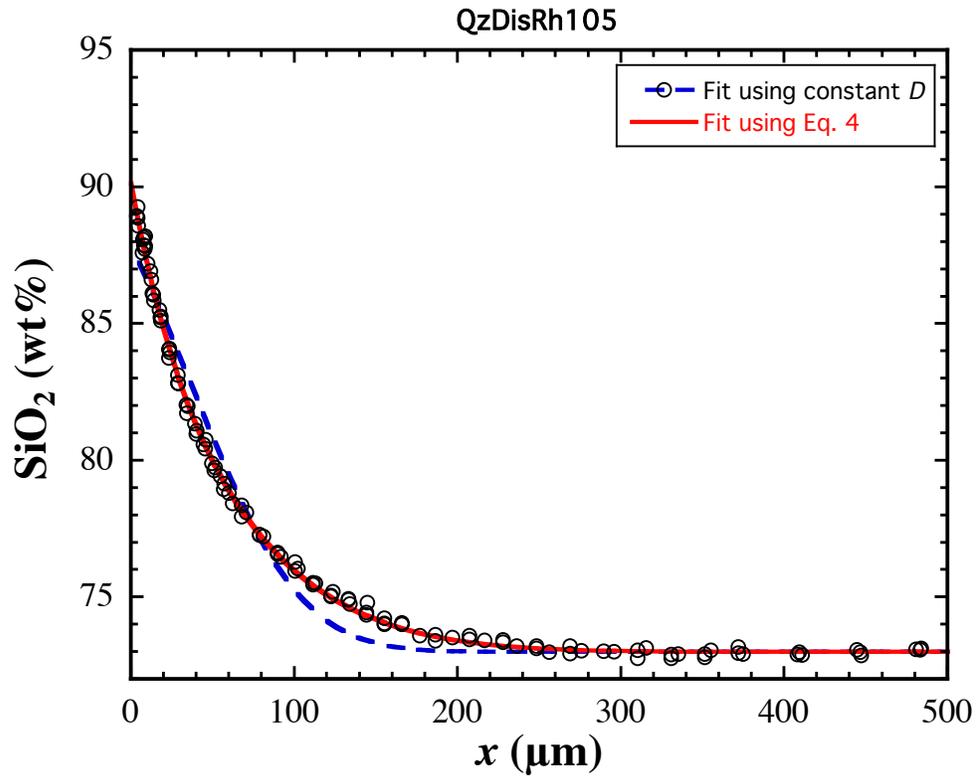
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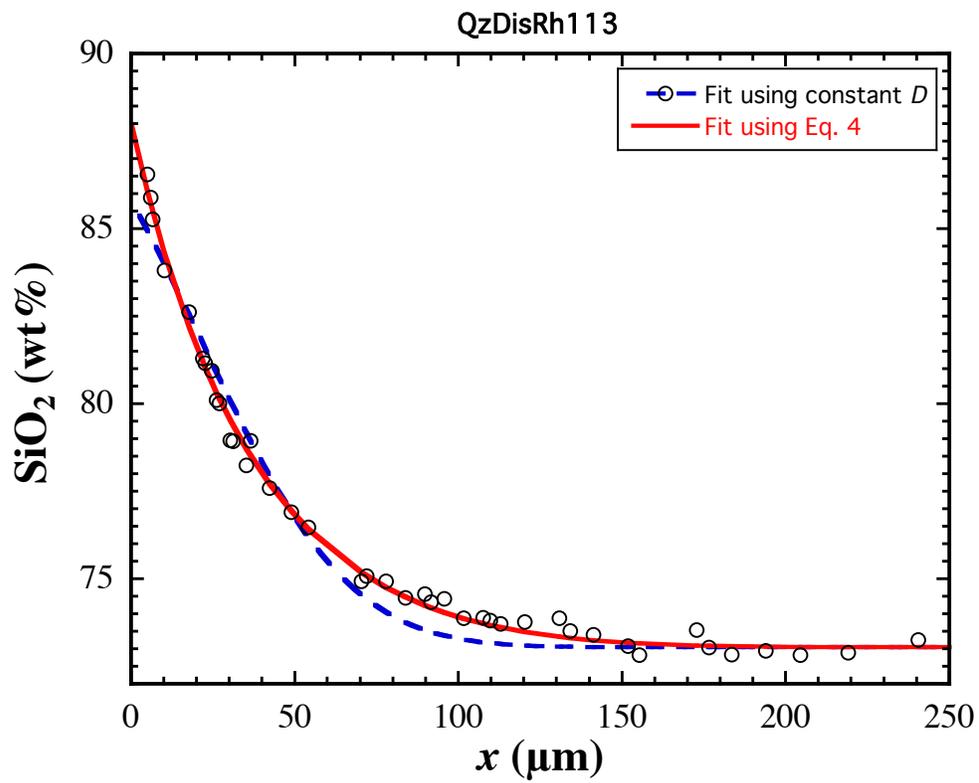
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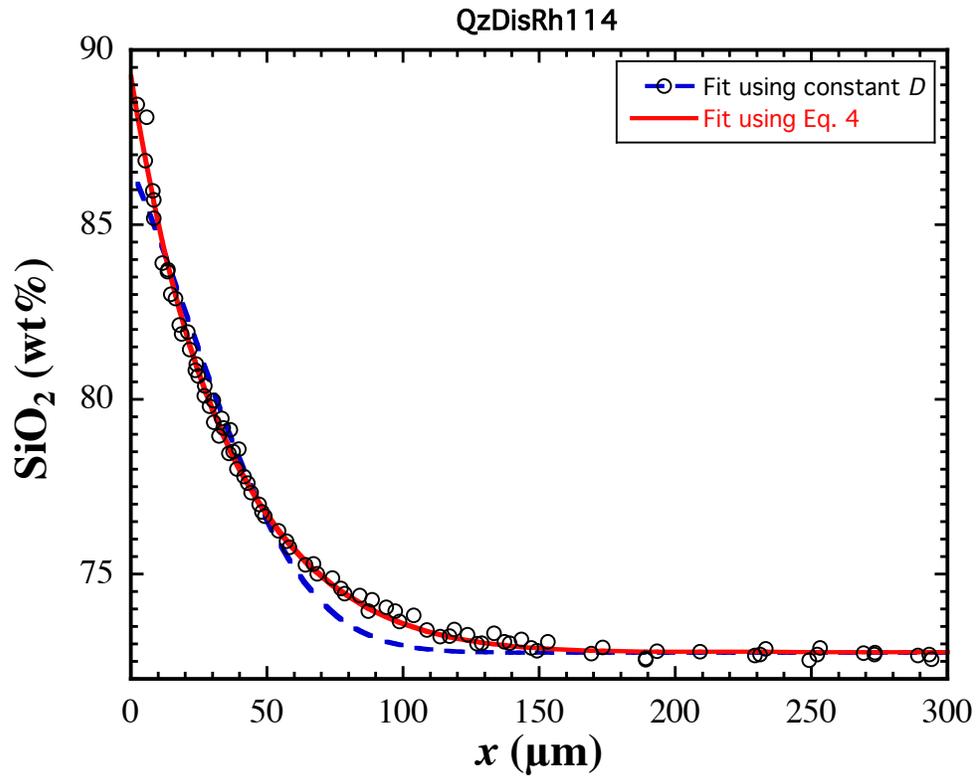
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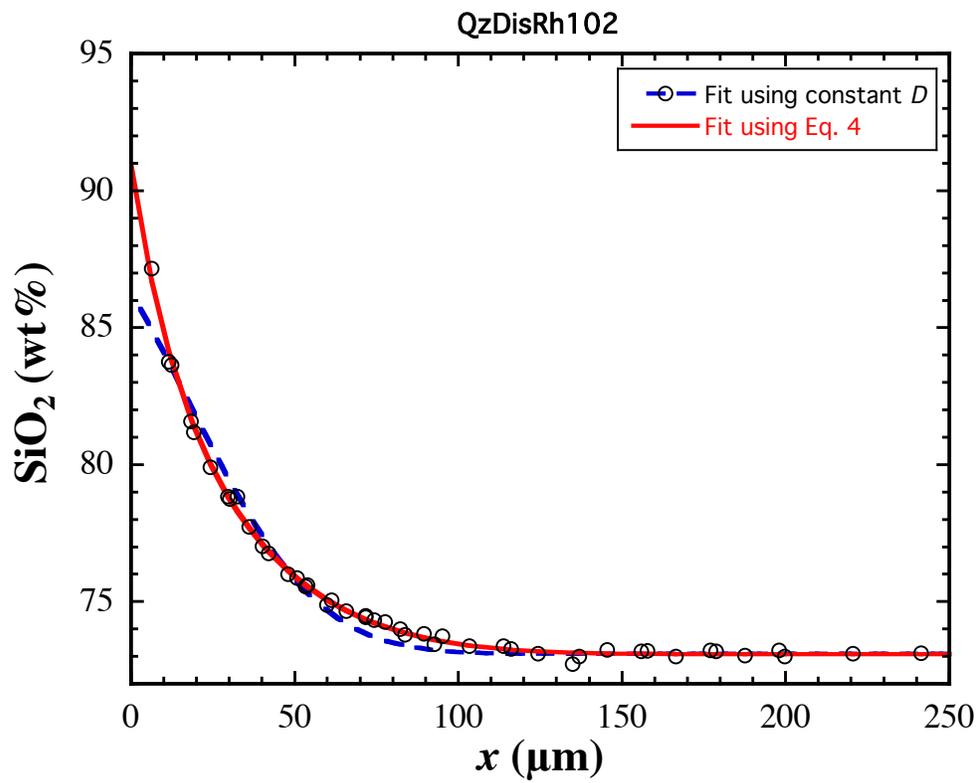
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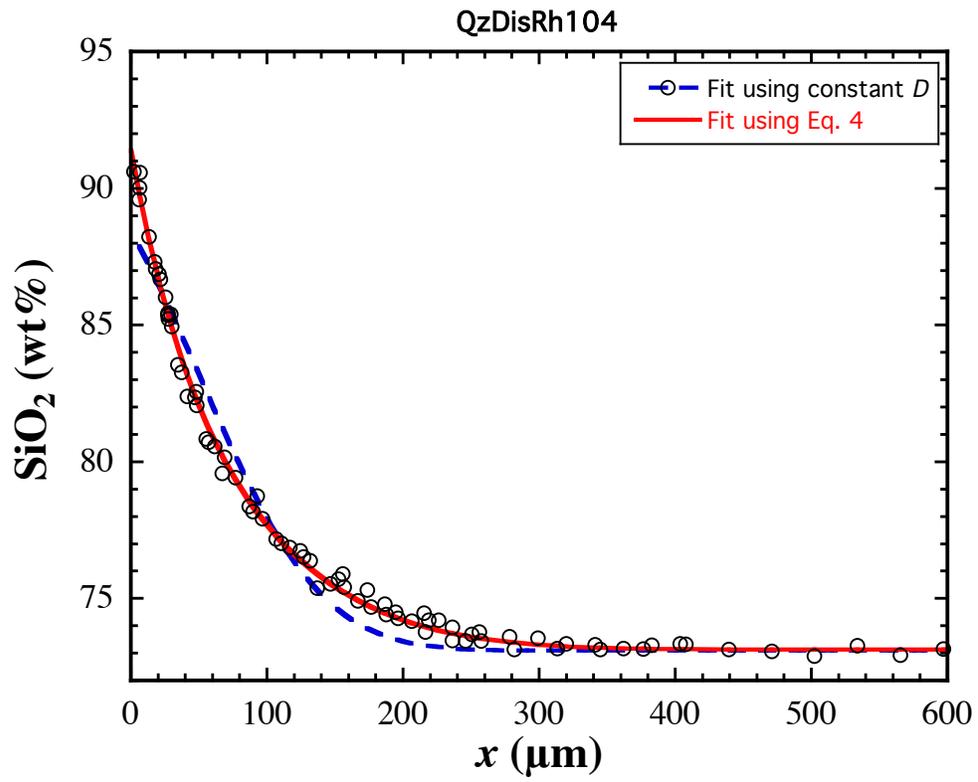
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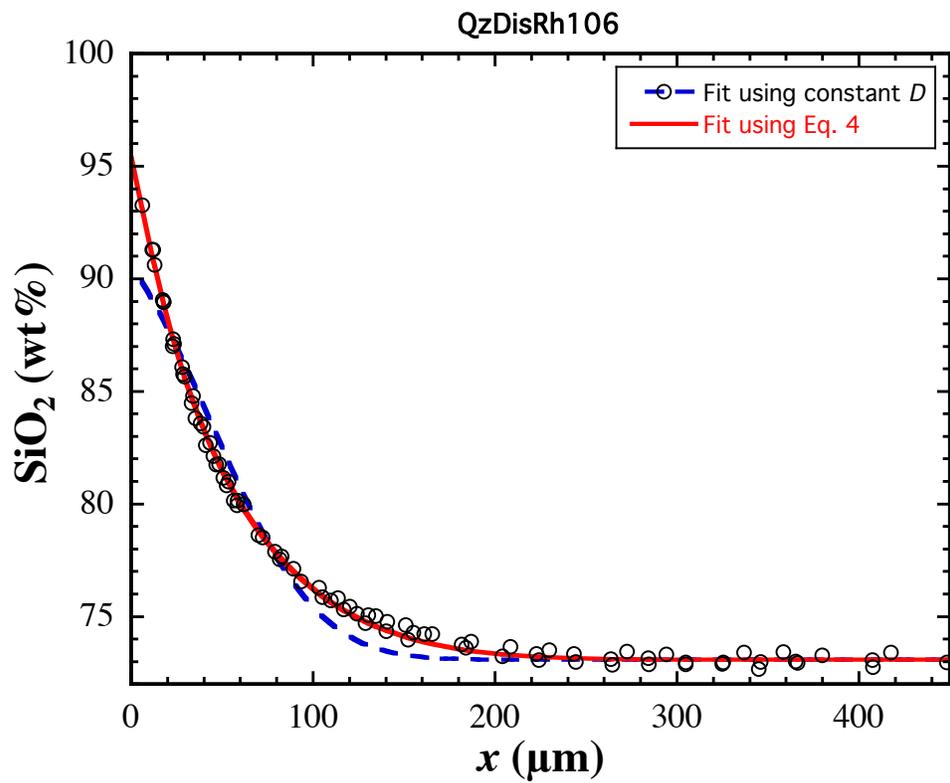
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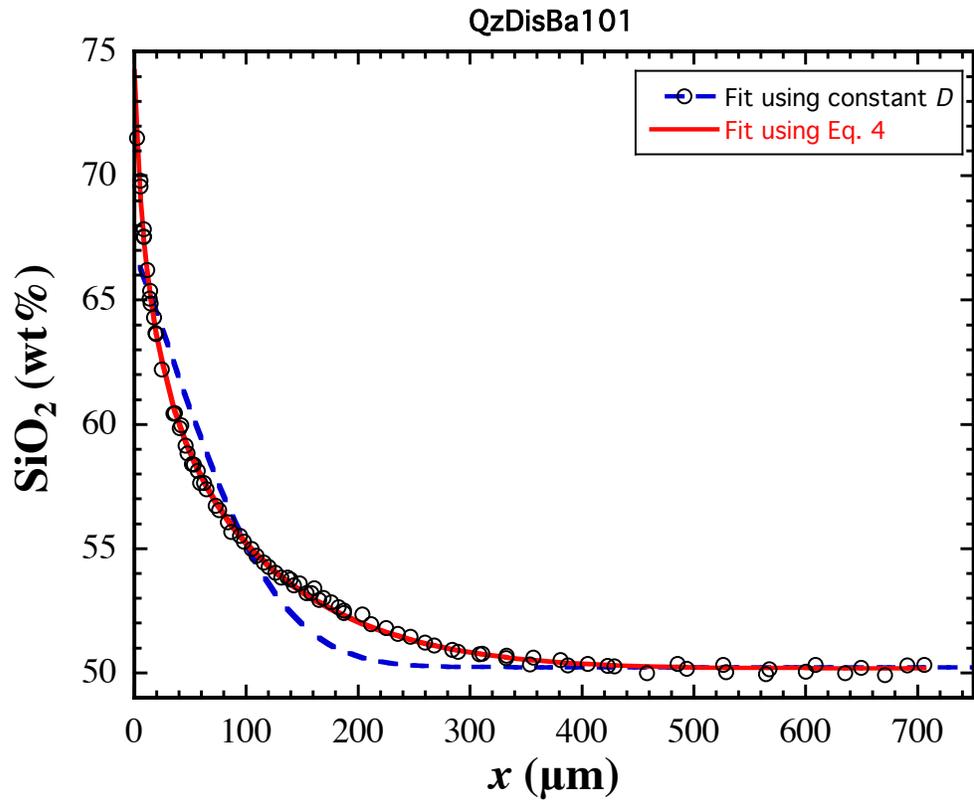
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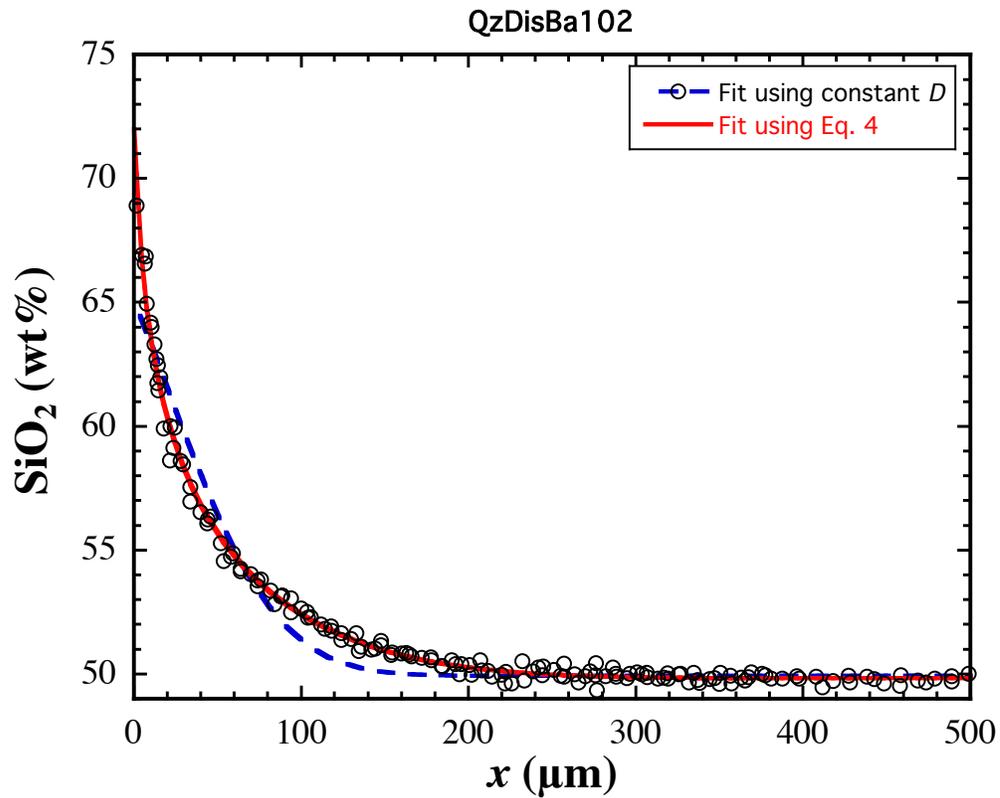
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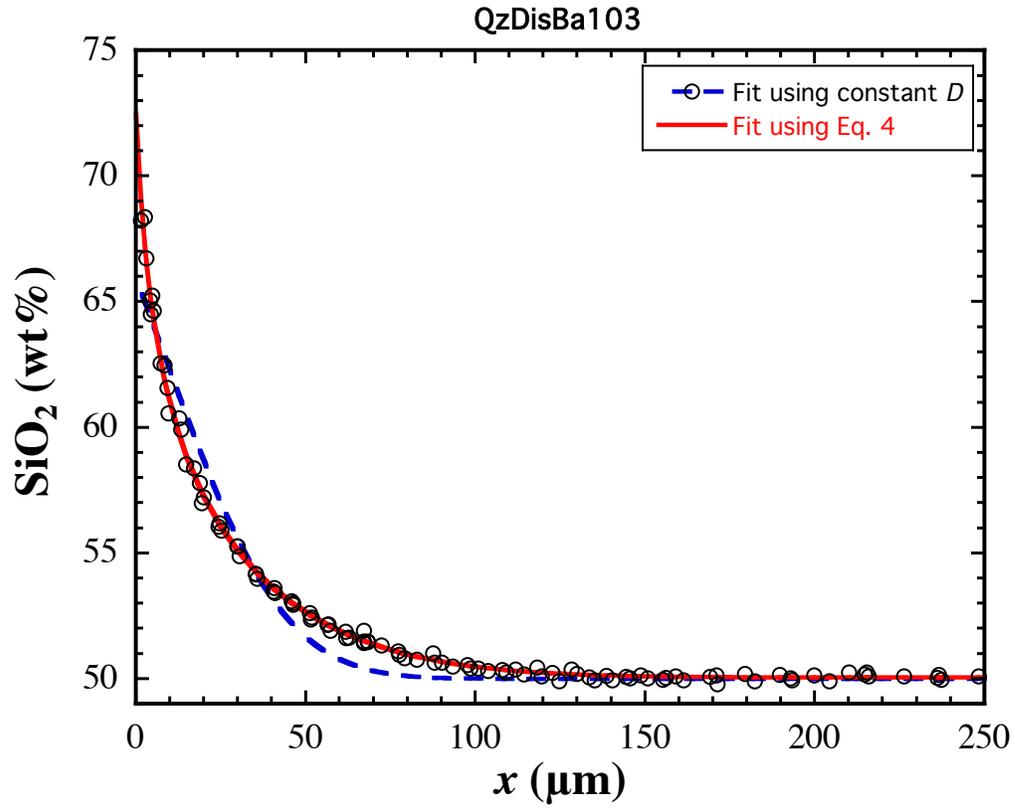
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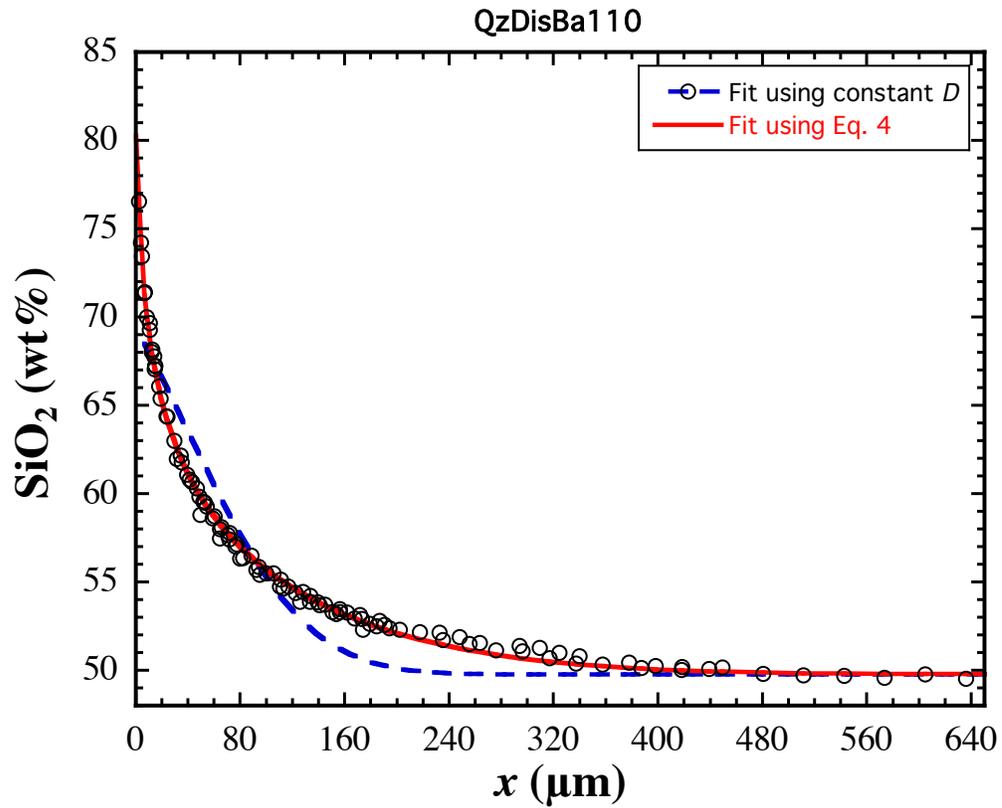
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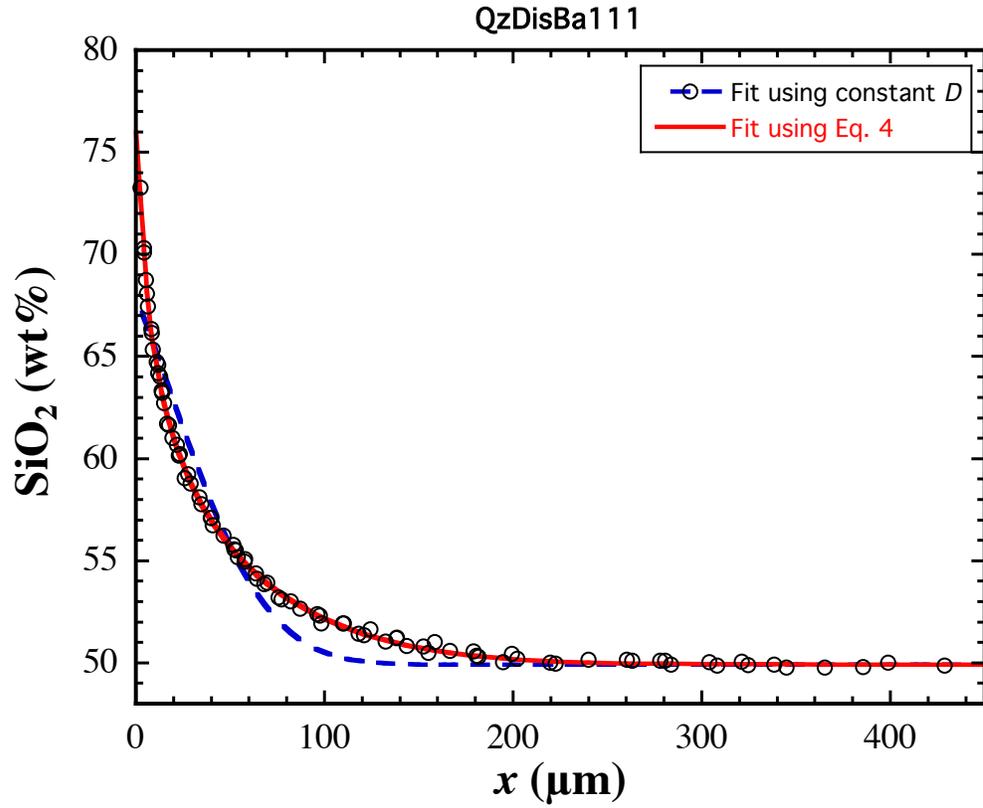
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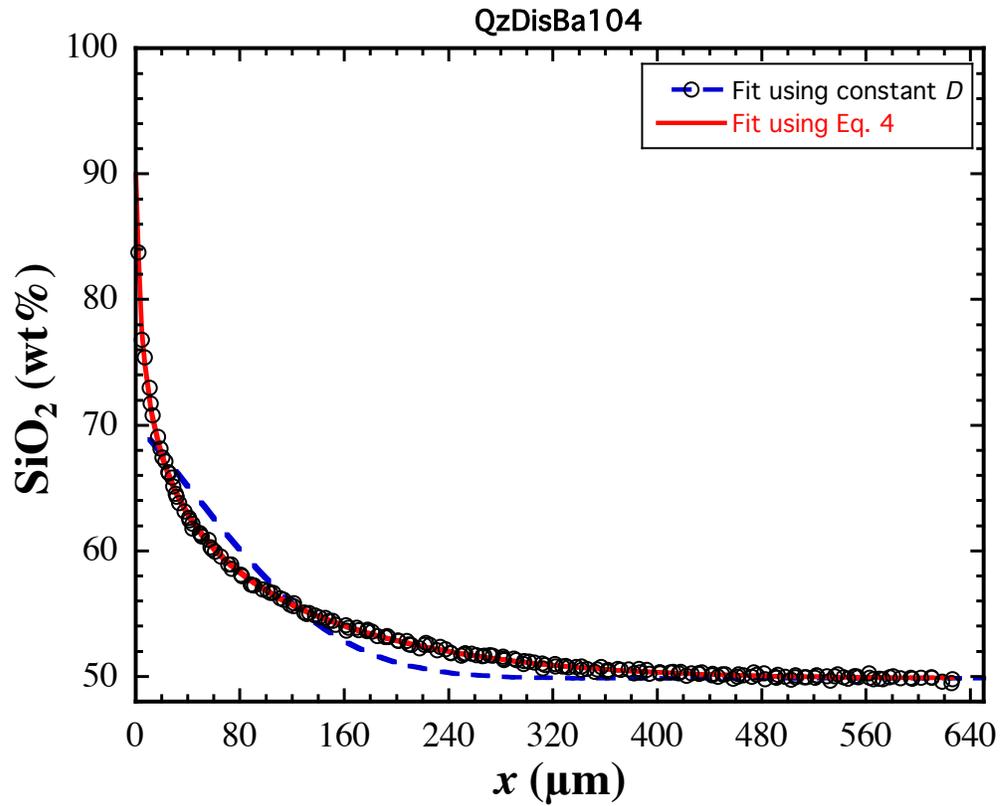
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QzDisBa104



QzDisBa107

QzDisBa107

